

# Oxide MBE— A Tool to Create Artificial Quantum Materials

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# Outline

- What is MBE and what is it good for?

*Greatest hits of MBE*

- How to grow your favorite oxide quantum material by MBE?

*Nuts and bolts of oxide MBE*

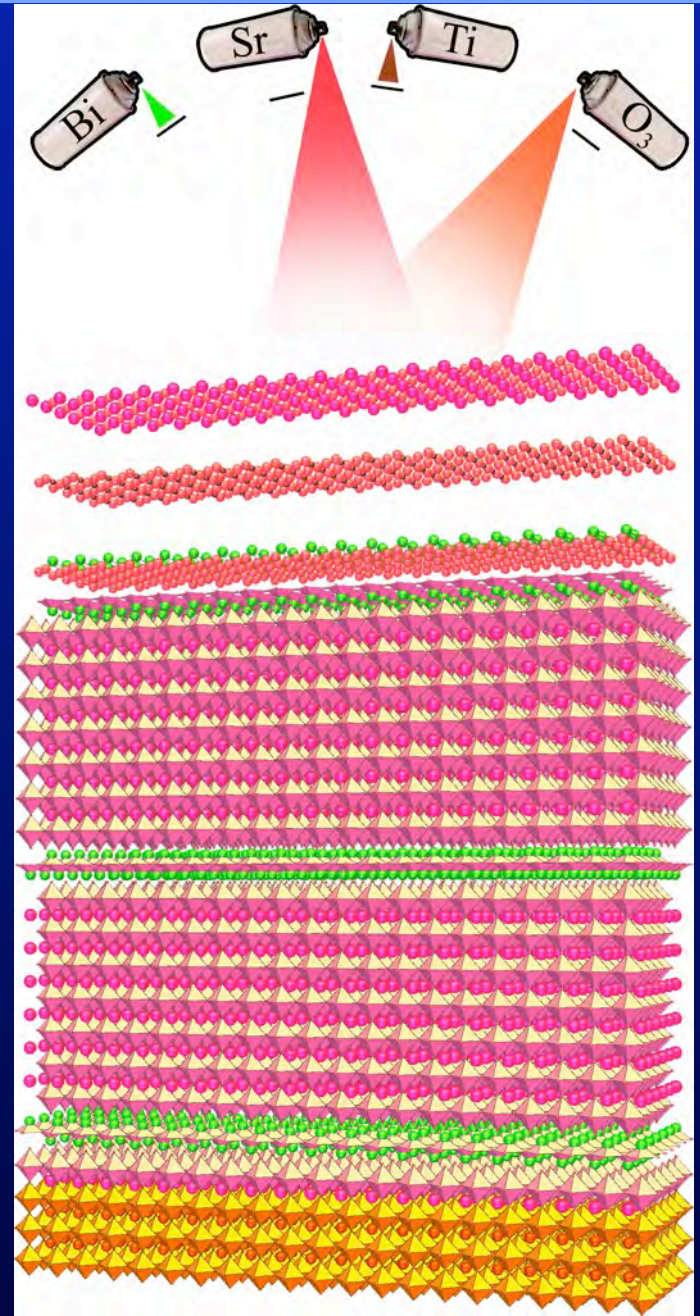
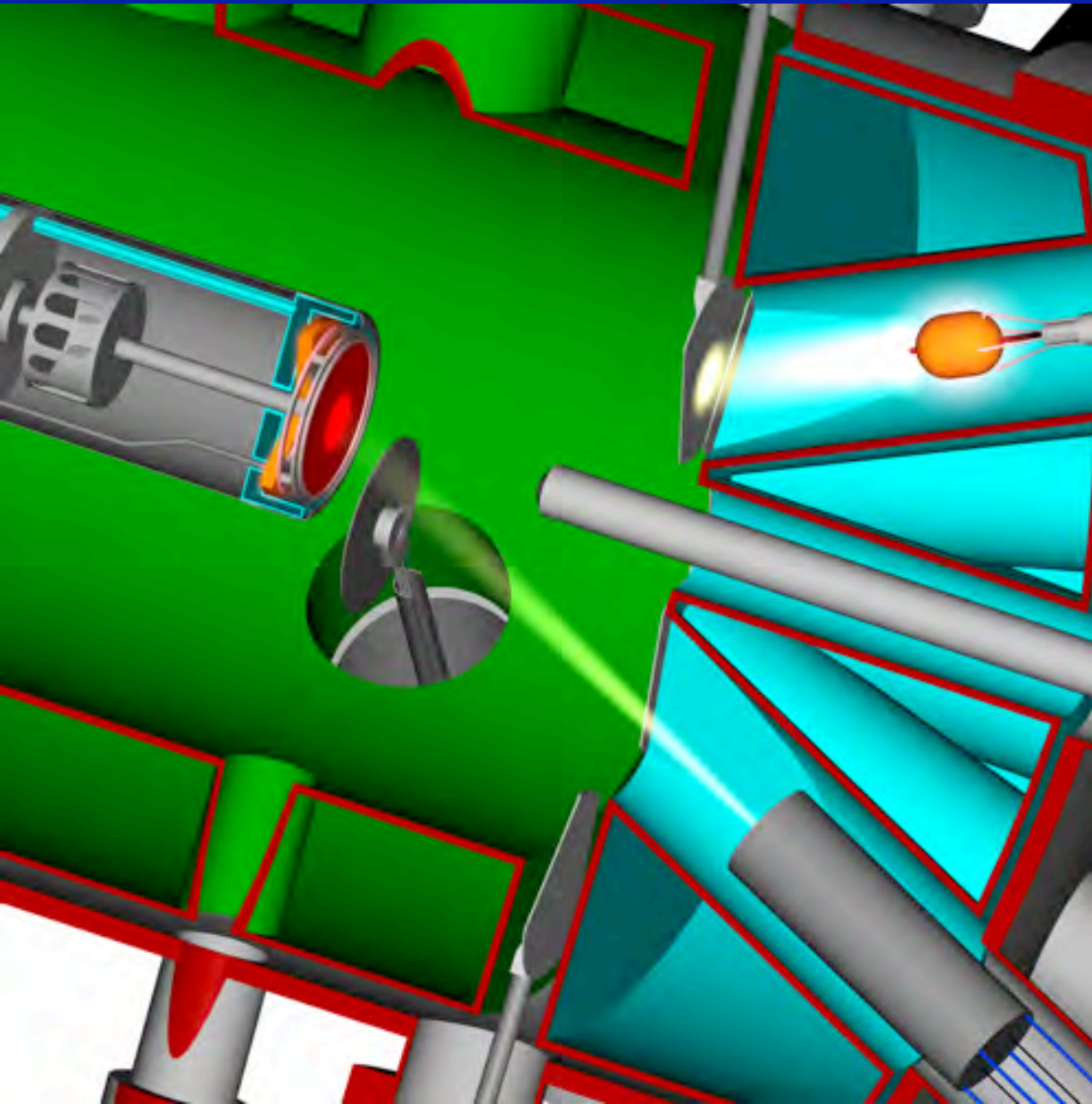
- Oxide MBE growth of quantum materials

*Case studies—including  $\text{Sr}_2\text{RuO}_4$*

- How can I gain access to an oxide MBE if I don't have one?

*Use PARADIM's oxide MBE (+ ARPES + ...)*

# MBE $\approx$ Atomic Spray Painting



# When GaAs is Heated ...

*J. Phys. Chem. Solids* Pergamon Press 1967. Vol. 28, pp. 2257-2267. Printed in Great Britain.

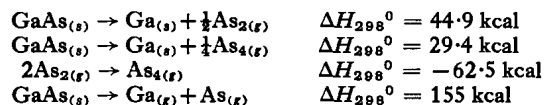
## VAPOR PRESSURES AND PHASE EQUILIBRIA IN THE Ga-As SYSTEM

J. R. ARTHUR

Bell Telephone Laboratories, Incorporated, Murray Hill, New Jersey

(Received 9 March 1967; in revised form 18 May 1967)

**Abstract**—Mass spectrometric and weight loss measurements of the species effusing from a Knudsen cell containing GaAs were used to obtain vapor pressures over the temperature range 900–1200°K. The  $As_2/As_4$  ratio was observed in these measurements to be substantially larger than previously reported<sup>(2,3)</sup> when precautions were taken to prevent the buildup of arsenic vapor in the mass spectrometer ionization chamber. A third law treatment of the data gave enthalpies for the reactions:



These results were used to correct Thurmond's calculations of vapor pressures and activity coefficients along the GaAs liquidus.<sup>(1)</sup>

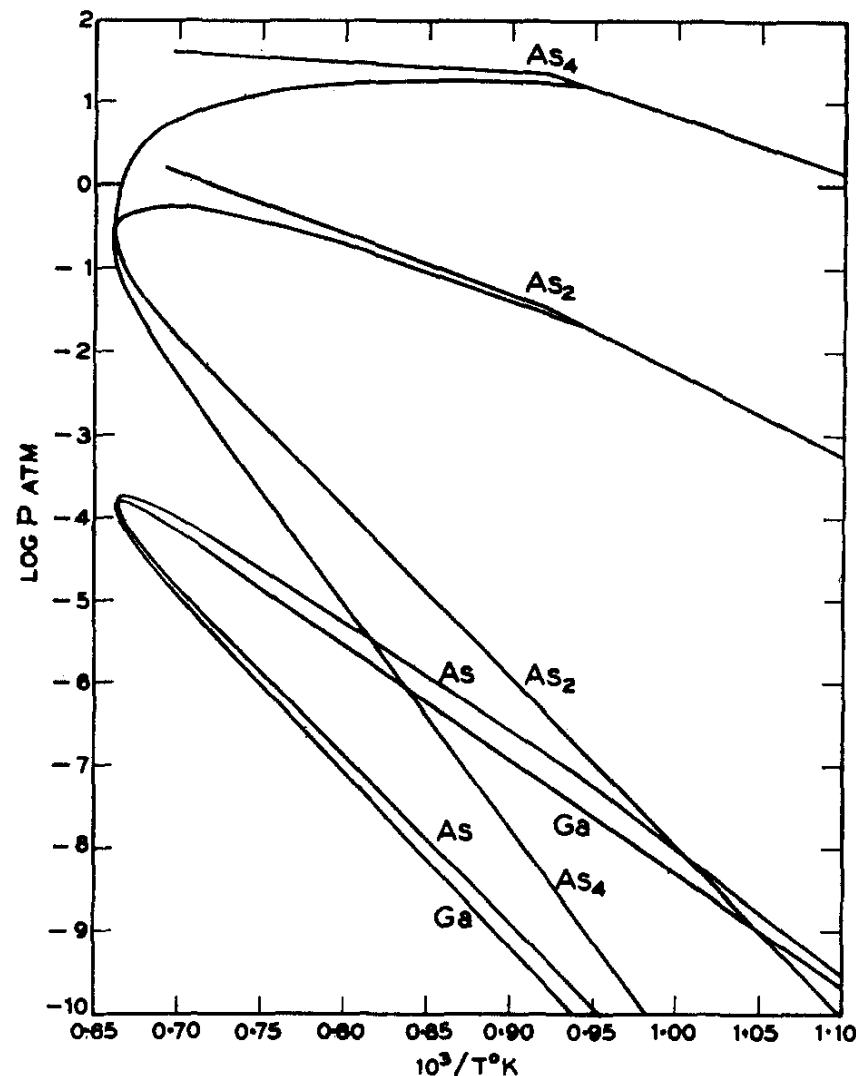


FIG. 5. Equilibrium vapor pressures of As,  $As_2$ ,  $As_4$  and Ga along the binary liquidus as a function of  $T^{-1}$ . Pressures of  $As_2$  and  $As_4$  over pure solid and liquid As are also shown.

# Consider Evaporation of PbO

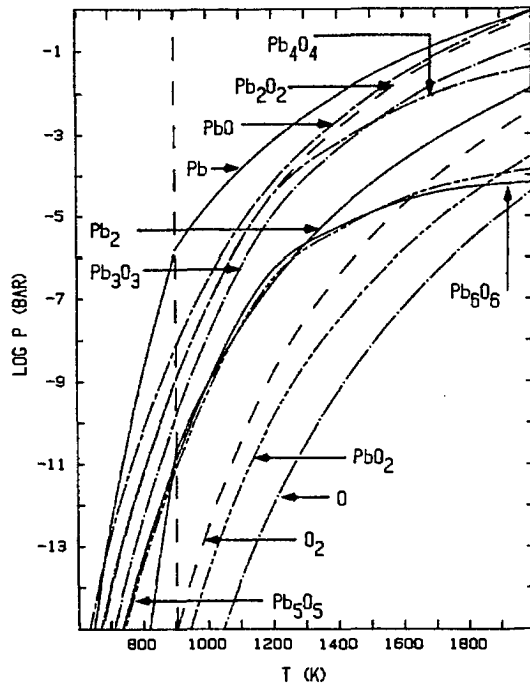


FIG. 53. PbO vaporization in  $10^{-15}$  bar  $O_2$  below 905 K and vaporization of Pb-PbO mixture above 905 K.

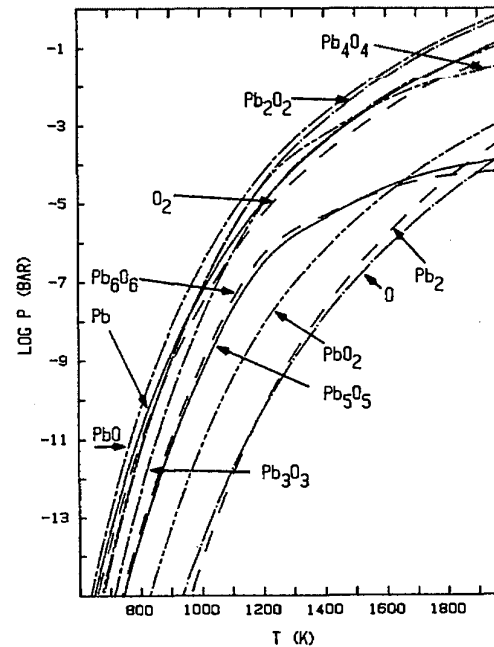


FIG. 54. PbO congruent vaporization.

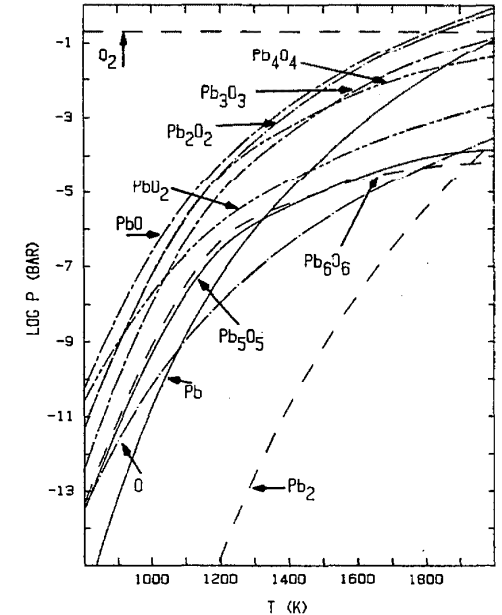


FIG. 55. PbO vaporization in 0.2 bar  $O_2$ .

R.H. Lamoreaux and D.L. Hildenbrand,

“High-Temperature Vaporization Behavior of Oxides II. Oxides of Be, Mg, Ca, Sr, Ba, B, Al, Ga, In, Tl, Si, Ge, Sn, Pb, Zn, Cd, and Hg,”

*J. Phys. Chem. Ref. Data* 16 (1987) 419-443.

# Key Enablers of MBE

- “3-Temperaturaufdampfverfahren” for Growth of III-V Semiconductor Films by Vacuum Evaporation  
K.G. Günther, “Aufdampfschichten aus halbleitenden III-V Verbindungen,” *Zeitschrift für Naturforschung A* **13** (1958) 1081-1089.
- Reliable UHV Sealing Technology  
W.R. Wheeler and M. Carlson, “Ultra-High Vacuum Flanges,” *Transactions of the Eighth National Vacuum Symposium*, edited by L.E. Preuss (Pergamon, New York, 1962), pp. 1309-1318.  
M.A. Carlson and W.R. Wheeler, “Metal Vacuum Joint,” U.S. Patent #3,208,758 (Sept. 28, 1965).

# Epitaxial GaAs by 3-Temperature-Technique

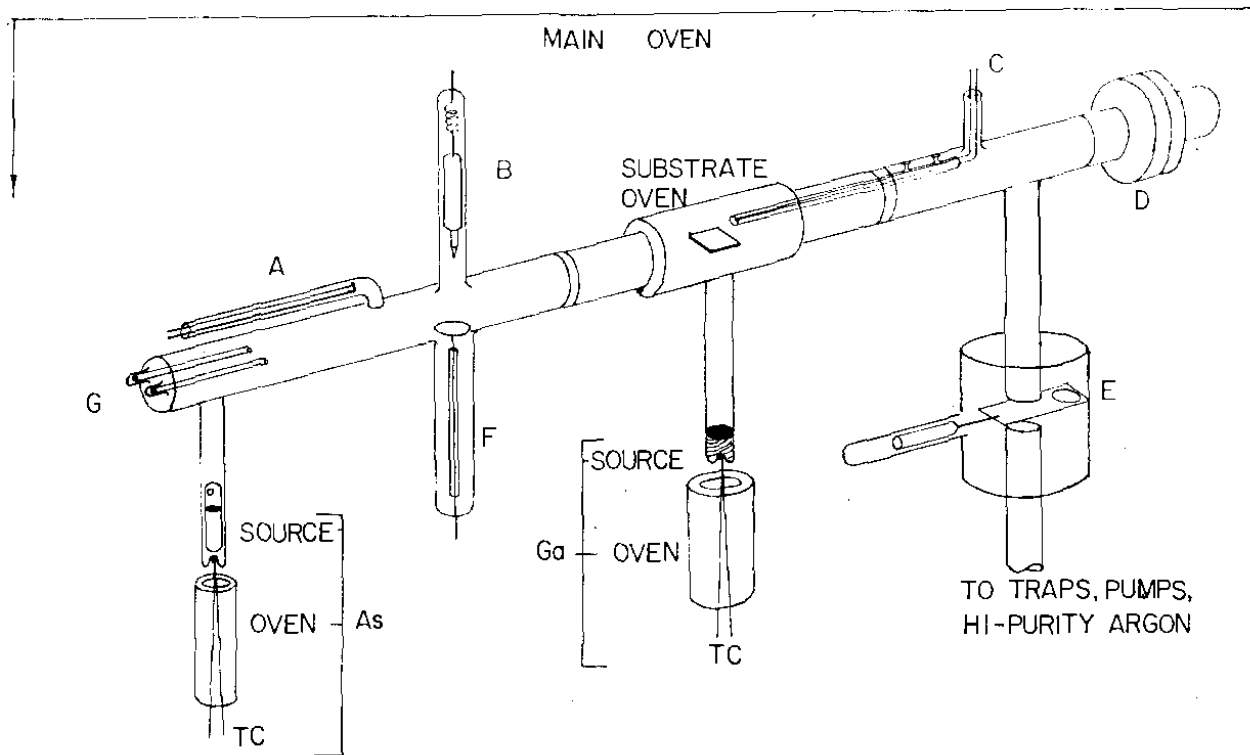


FIG. 1. GaAs film evaporation system: (A) Pirani gauge; (B) electrical contact to diode structure; (C) thermocouple; (D) metal flanges and viton gaskets as an entrance port for loading system; (E) particulate valve; (F) circular Ta plate; positive electrode in diode structure; (G) quartz rods which extend the length of the envelope and which guide the substrate carrier.

# UHV Seals—Varian Conflat®

Sept. 28, 1965

M. A. CARLSON ET AL

3,208,758

METAL VACUUM JOINT

Filed Oct. 11, 1961

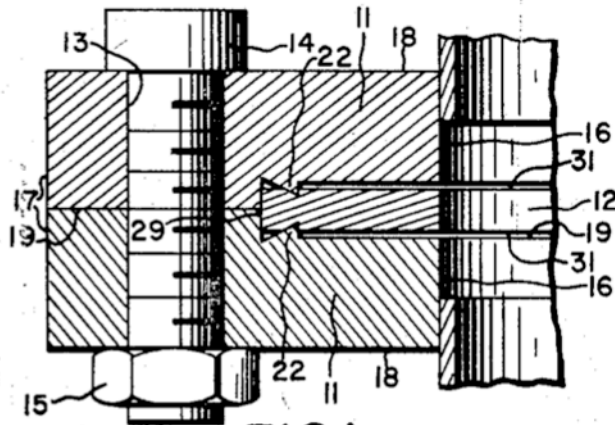


FIG. 1

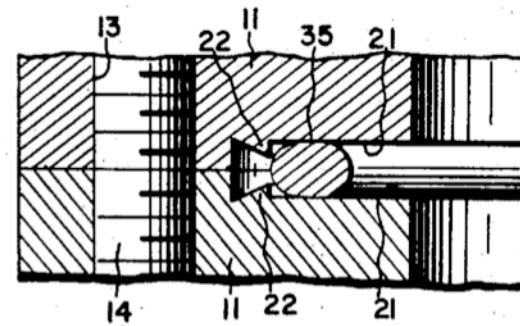


FIG. 3

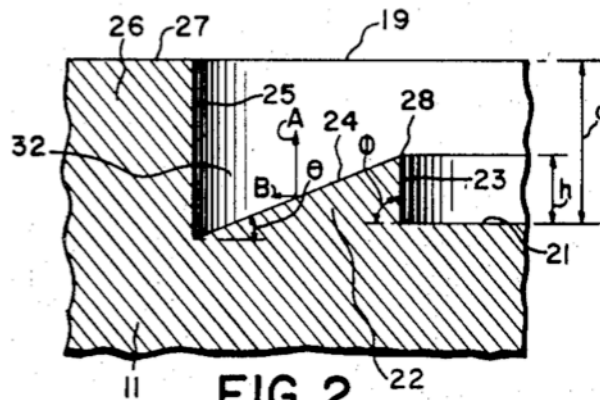


FIG. 2

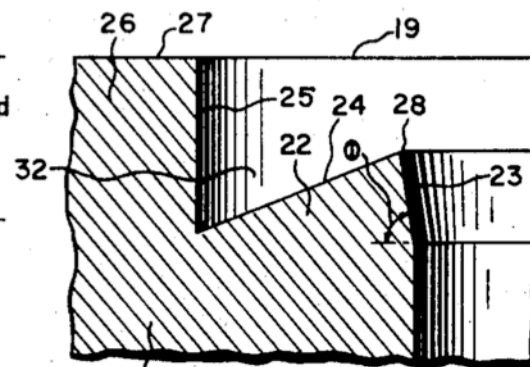


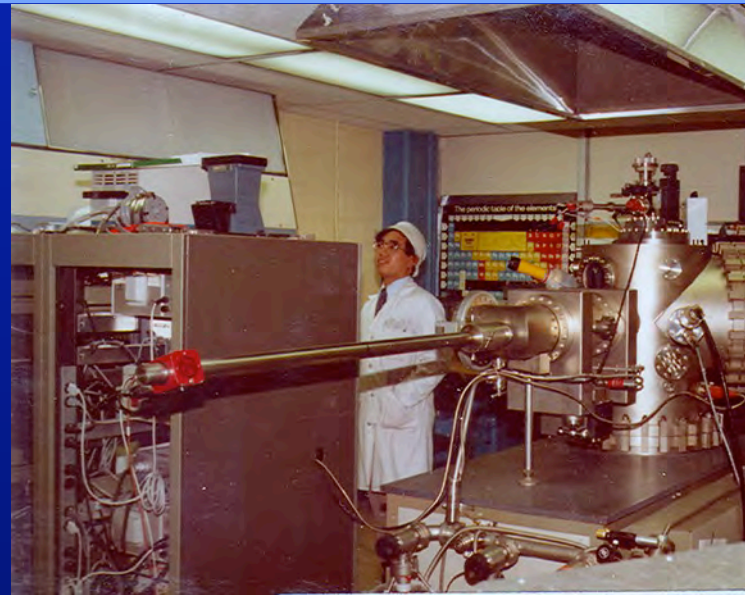
FIG. 4



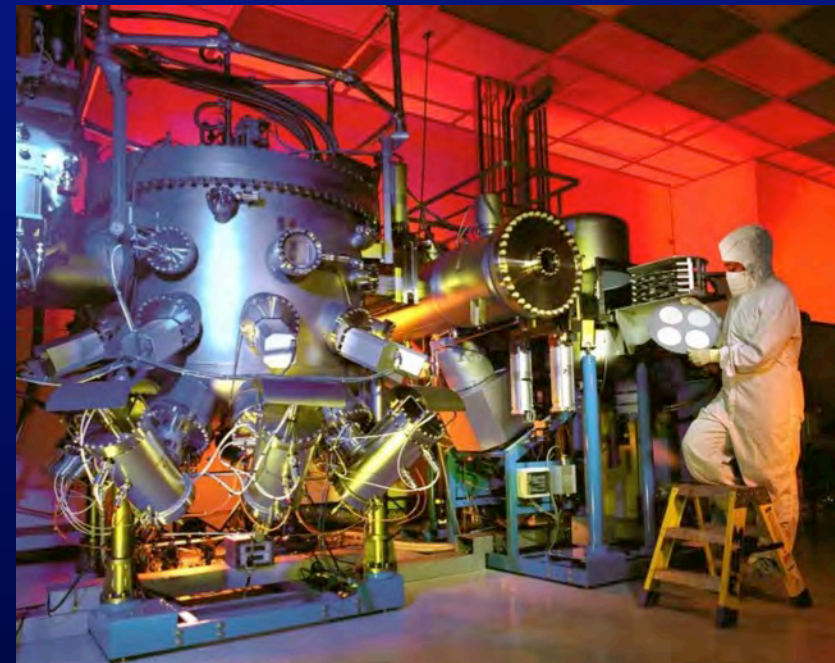
# Evolution of MBE



1<sup>st</sup> MBE  
Al Cho at Bell Labs, 1972



1<sup>st</sup>  
University  
MBE  
Cornell,  
1978



Production  
MBE  
Today  
(courtesy of TRW)

# MBE production tool performance data

## HIGH YIELD

### UNIFORMITIES / Wafer

Thickness <  $\pm 0.5$  %

Composition <  $\pm 0.5$  %

Doping <  $\pm 1$  %

### REPRODUCIBILITY

Source material: supply consistency

Stable process and monitoring: < 2%

## HIGH THROUGHPUT

### VERY HIGH UPTIME

> 94%, run 6 to 9 months, 7 days/wk, 24/24

### RUN CAPABILITY

13x2'' or 5x3'', 4x6'' or 9x4'', (4x8'') 7x6''

### RUN SWITCHING

less than 2 minutes (platen exchange)

# Reflection High-Energy Electron Diffraction (RHEED) Oscillations

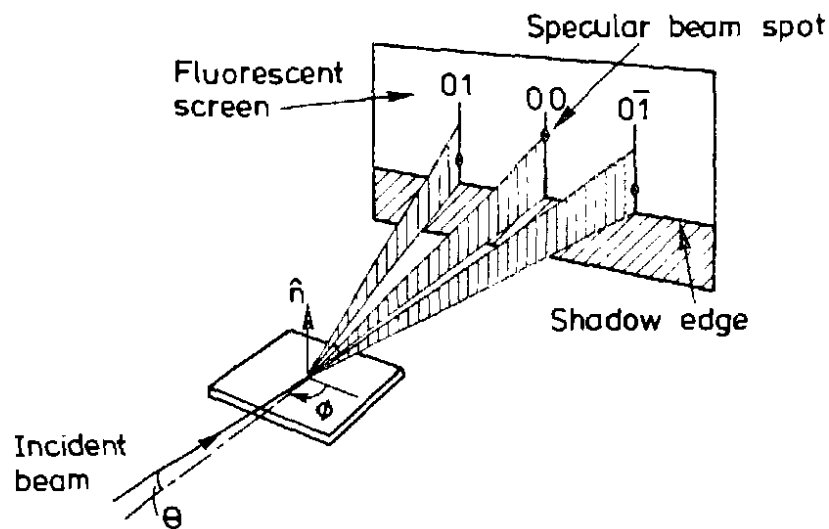
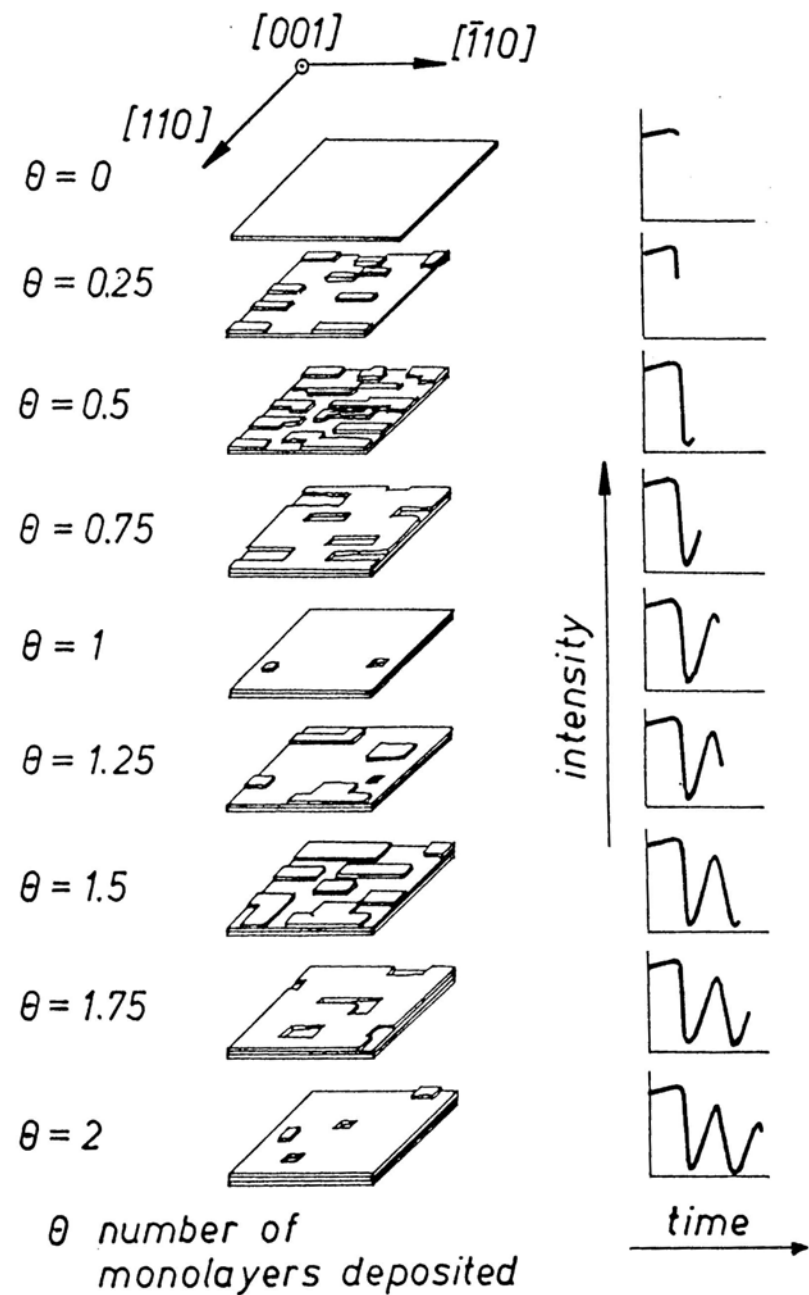


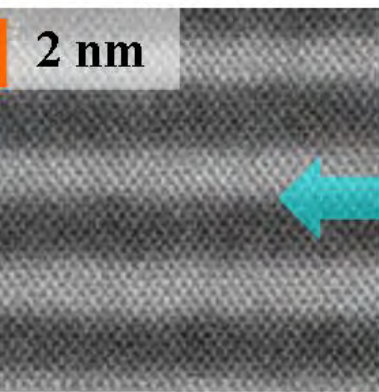
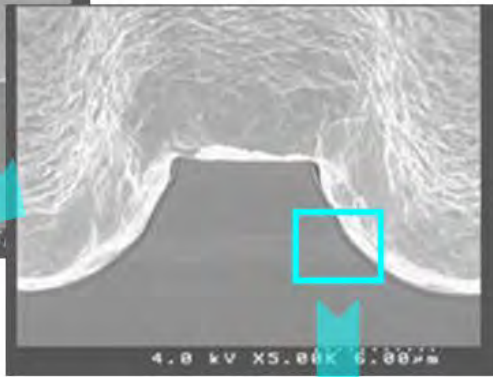
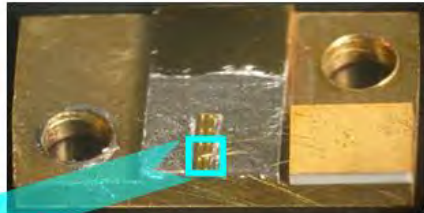
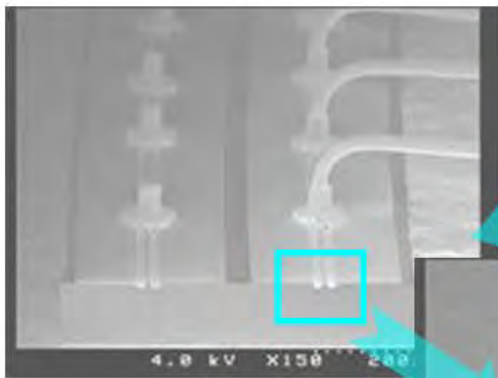
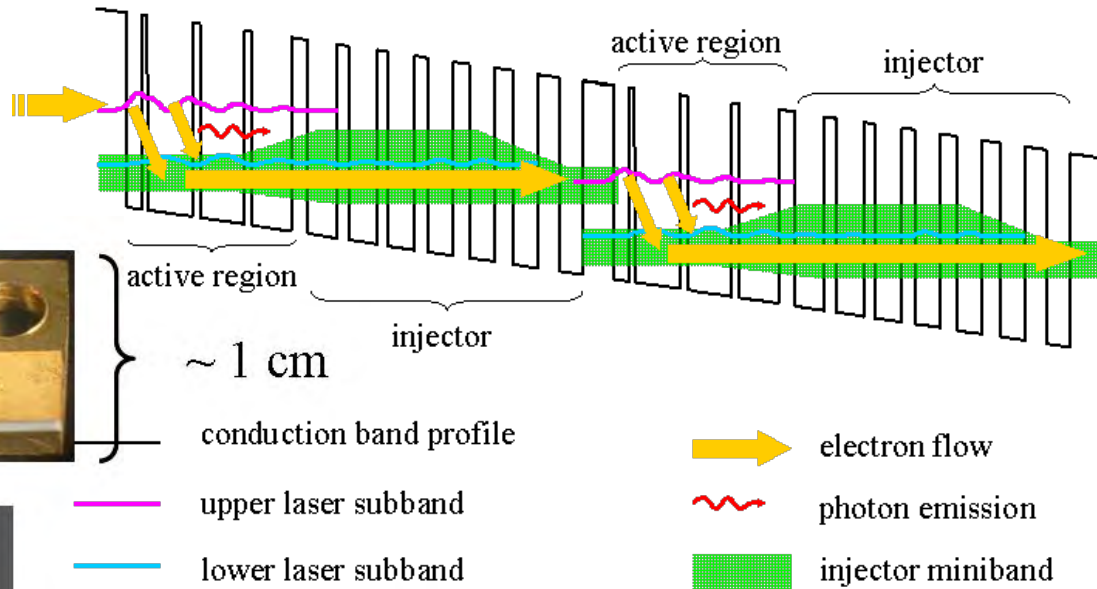
FIG. 1. Schematic diagram of RHEED geometry showing the incident beam at an angle  $\theta$  to the surface plane; azimuthal angle  $\phi$ . The elongated spots indicate the intersection of the Ewald sphere with the  $01$ ,  $00$ , and  $0\bar{1}$  rods.



B. Bölger and P. K. Larsen, *Review of Scientific Instruments* **57** (1986) 1363-1367.

B.A. Joyce, P.J. Dobson, J.H. Neave, K. Woodbridge, J. Zhang, P.K. Larsen, and B Bölger, *Surface Science* **168** (1986) 423-438.

# Quantum Cascade Laser



# What is MBE?

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- (a) Molecular-Beam Epitaxy
- (b) Mega-Buck Evaporator
- (c) Many Boring Evenings
- (d) Mainly Broken Equipment
- (e) All of the above

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- How to grow your favorite oxide quantum material by MBE?

*Nuts and bolts of oxide MBE*

- Oxide MBE growth of quantum materials

*Case studies—including  $Sr_2RuO_4$*

- How can I gain access to an oxide MBE if I don't have one?

*Use PARADIM's oxide MBE (+ ARPES + ...)*

# MBE for Science / Technology

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- 1998 Nobel Prize in Physics—  
Fractional Quantum Hall Effect
  - Horst Ludwig Störmer
  - Daniel Chee Tsui
  - Robert B. Laughlin
- 2000 Nobel Prize in Physics—  
Semiconductor Optoelectronics
  - Zhores Ivanovich Alferov
  - Herbert Kroemer

# Modulation Doping



R. Dingle, H.L. Störmer, A.C. Gossard, and W. Wiegmann, *Applied Physics Letters* **33** (1978) 665-667.

**Figure 2** Four pioneers of modulation doping gather around an early MBE machine at Bell Labs in 1978: (left–right) Willy Wiegmann, Art Gossard, Horst Störmer and Ray Dingle. Störmer and his Bell Labs colleague Daniel Tsui shared the Nobel prize for discovering the fractional quantum Hall effect in devices made by Gossard and co-workers with MBE.

W.P. McCray, *Nature Nanotechnology* **2** (2007) 259-261.



# Modulation Doping

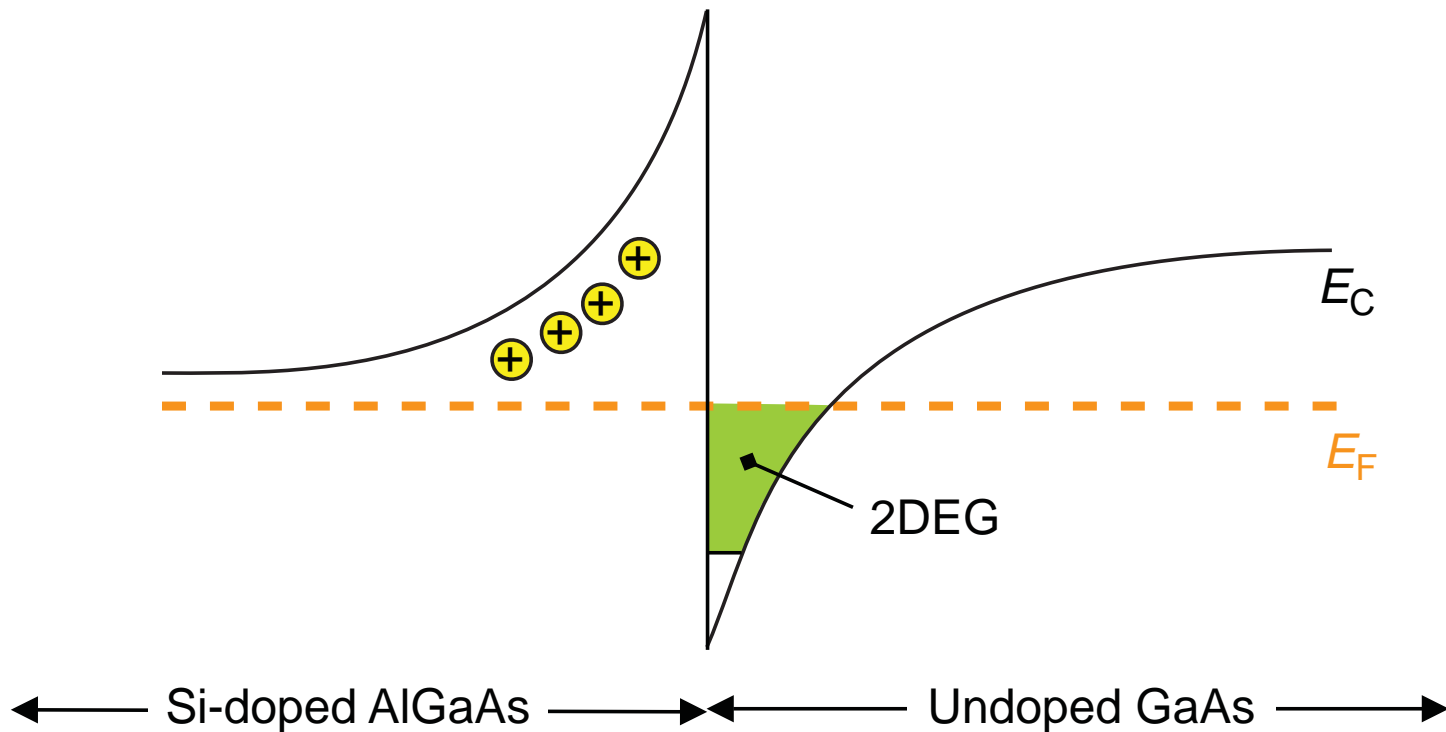
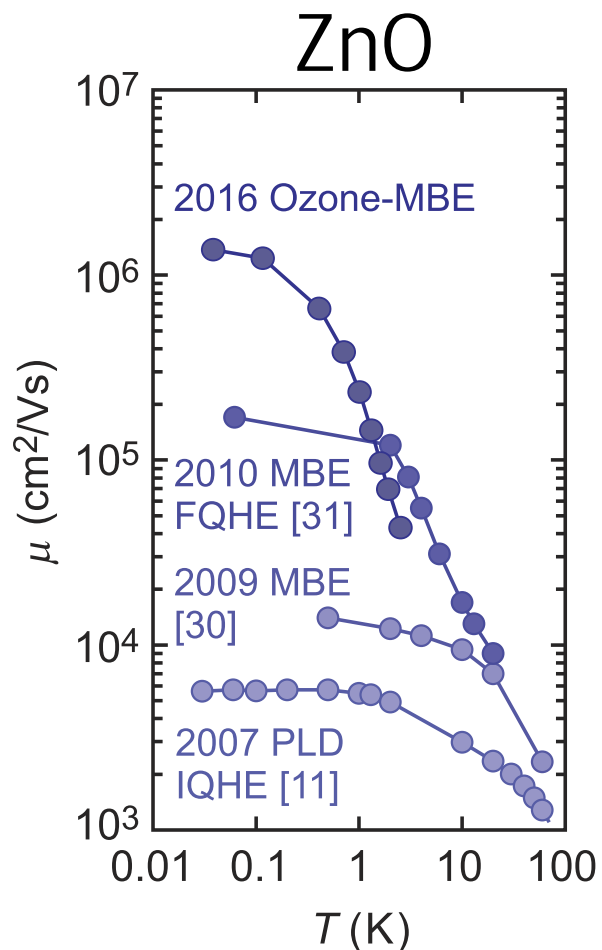


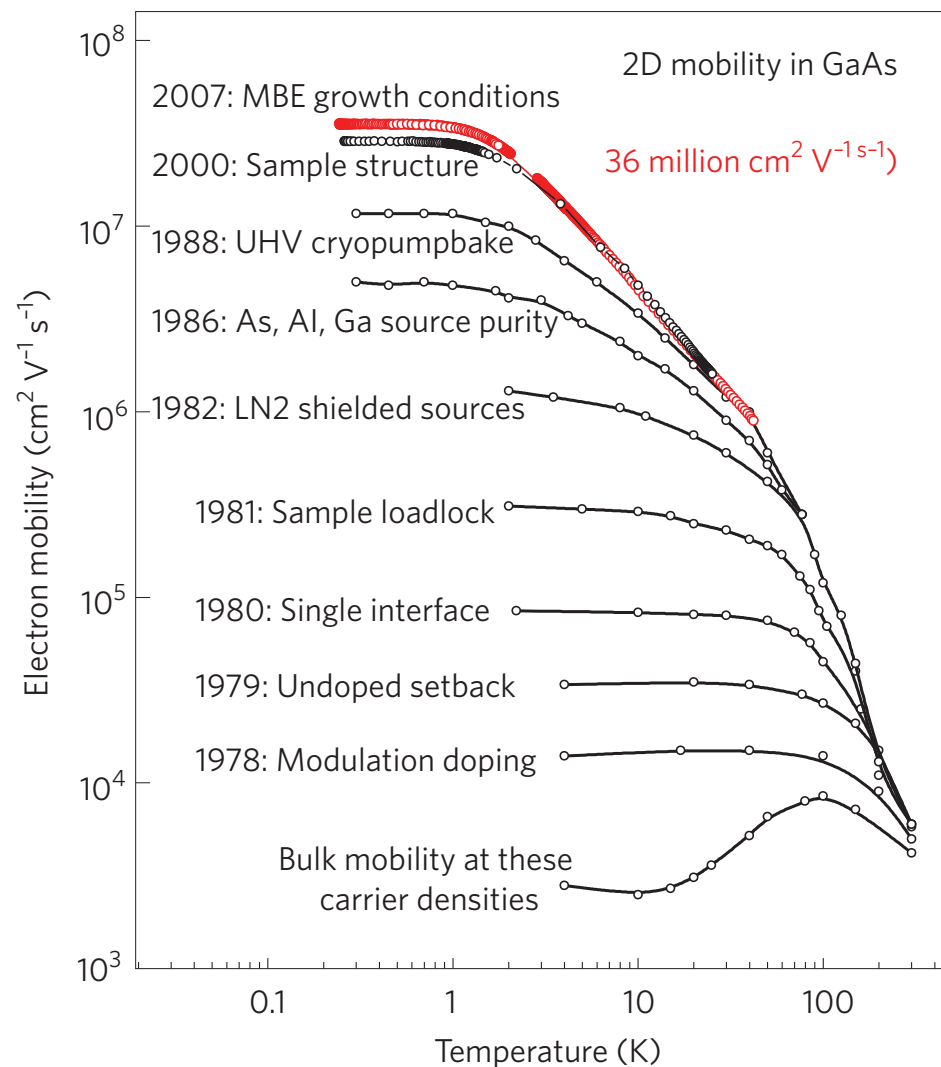
Figure 1. Band diagram showing the formation of a two-dimensional electron gas (2DEG) at a Si-doped AlGaAs–GaAs heterojunction. *Note:*  $E_F$  is the value of the Fermi energy, and  $E_C$  gives the energy of the conduction band edge.

# Mobility Achieved with MBE

$$m_{e, \text{ZnO}}^* > 4m_{e, \text{GaAs}}^*$$



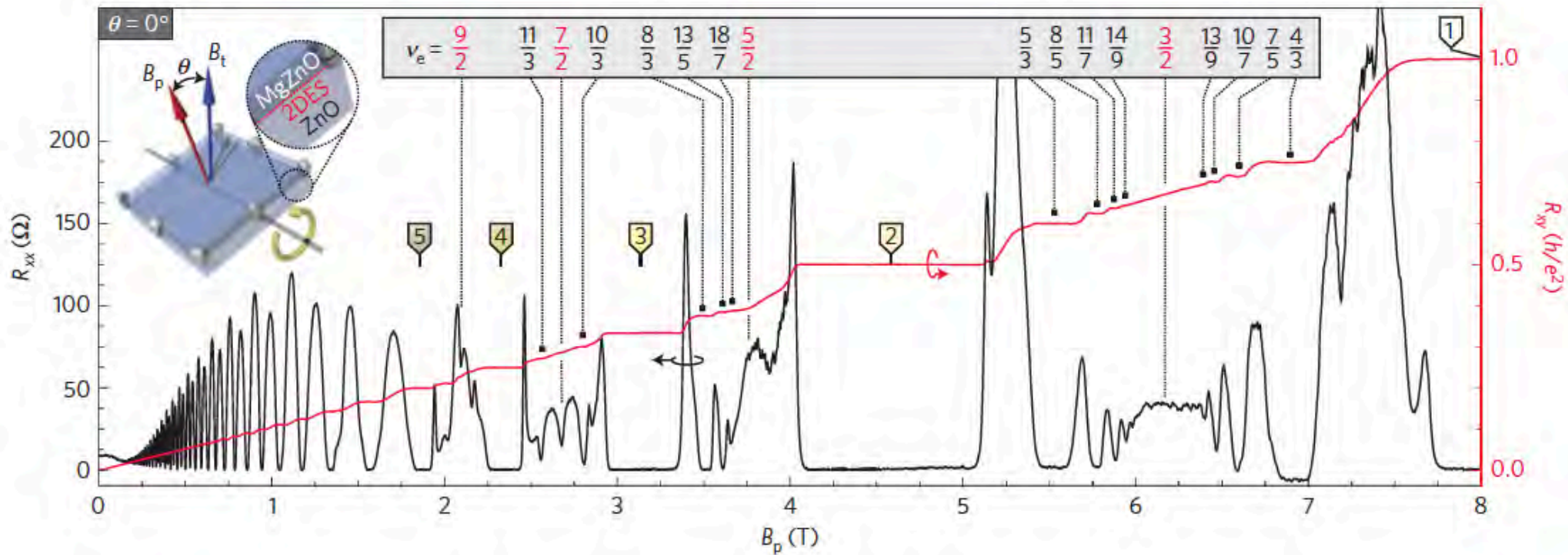
J. Falson, Y. Kozuka, M. Uchida, J. H. Smet, T.-H. Arima, A. Tsukazaki, and M. Kawasaki, *Scientific Reports* **6** (2016) 26598.



L. Pfeiffer and K.W. West, *Physics E* **20** (2003) 57-64.

D.G. Schlom and L.N. Pfeiffer, *Nature Materials* **9** (2010) 881-883.

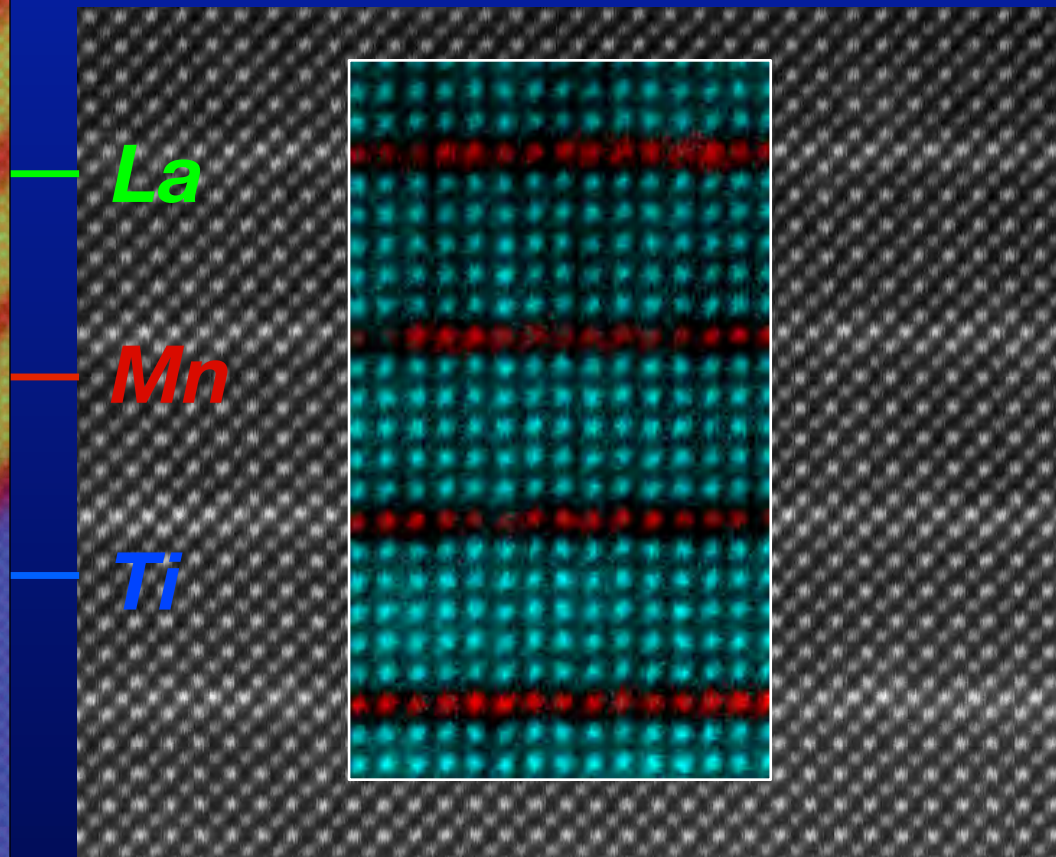
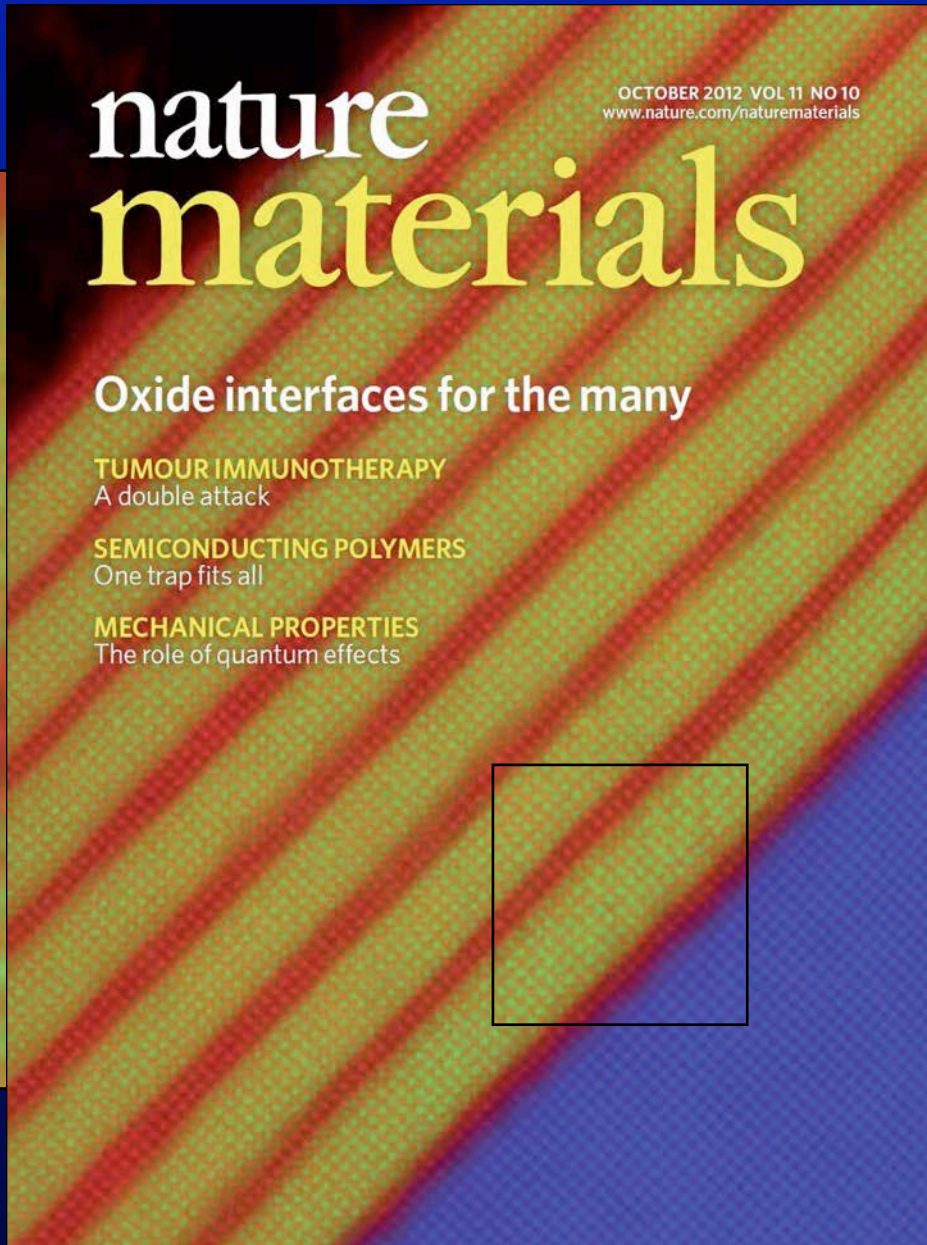
# Mobility Achieved with MBE



J. Falson, D. Maryenko, B. Friess, D. Zhang, Y. Kozuka, A. Tsukazaki, J. H. Smet, and M. Kawasaki,  
*Nature Physics* **11** (2015) 347–351.

# MBE of Quantum Materials

$(\text{SrRuO}_3)_1 / (\text{SrTiO}_3)_5$   
Superlattice



2 nm

Red = Ru

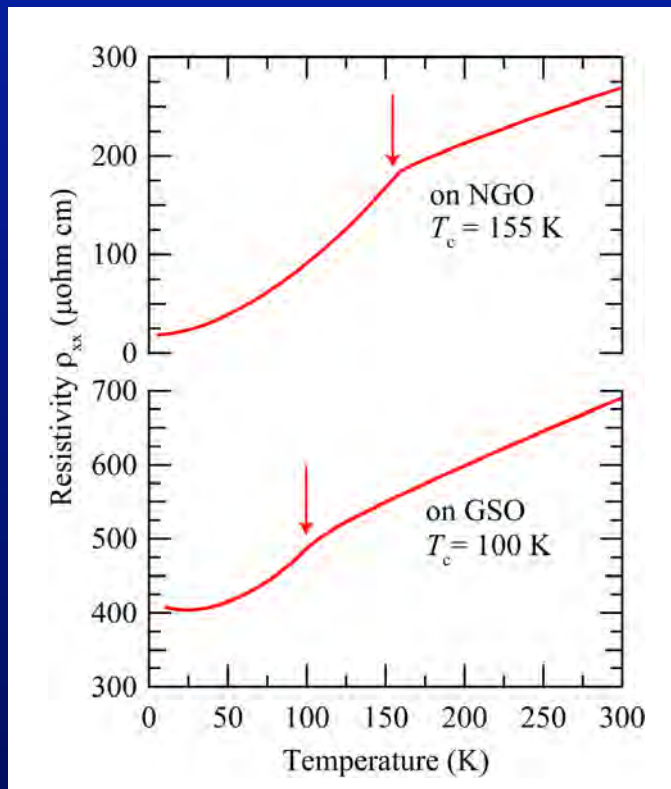
Teal = Ti

E.J. Monkman, C. Adamo, J.A. Mundy, D.E. Shai, J.W. Harter,  
D. Shen, B. Burganov, D.A. Muller, D.G. Schlom, and K.M. Shen,  
*Nature Materials* **11** (2012) 855-859.

# Transport of SrRuO<sub>3</sub> Films

Best PLD Film

$$\rho_{300\text{ K}} / \rho_{10\text{ K}} = 14.1$$

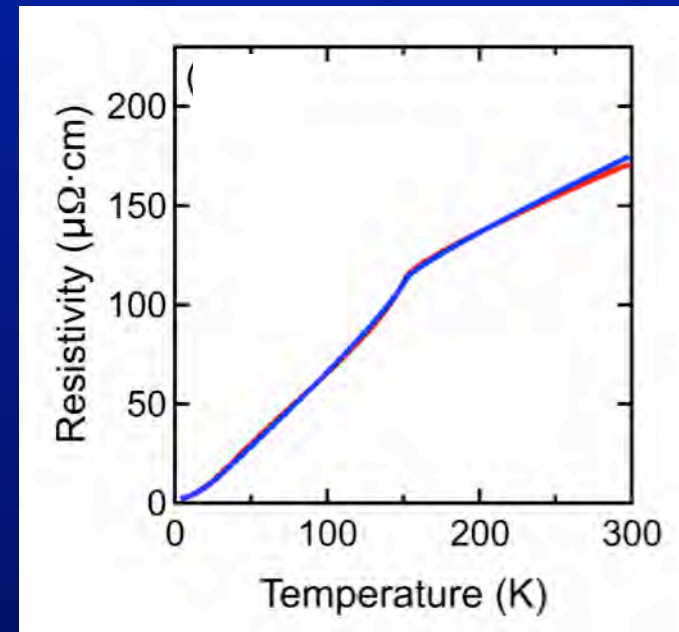


~20 nm SrRuO<sub>3</sub> / (110) NdGaO<sub>3</sub>

D. Kan, R. Aso, H. Kurata, and Y. Shimakawa,  
*J. Appl. Phys.* **113** (2013) 173912 .

Best MBE Film

$$\rho_{300\text{ K}} / \rho_{4\text{ K}} = 76$$



32 nm SrRuO<sub>3</sub> / (100) SrTiO<sub>3</sub>

H.P. Nair, Y. Liu, J.P. Ruf, N.J. Schreiber, S-L. Shang,  
D.J. Baek, B.H. Goodge, L.F. Kourkoutis,  
Z.K. Liu, K.M. Shen, and D.G. Schlom,  
*APL Materials* **6** (2018) 046101.

# Outline

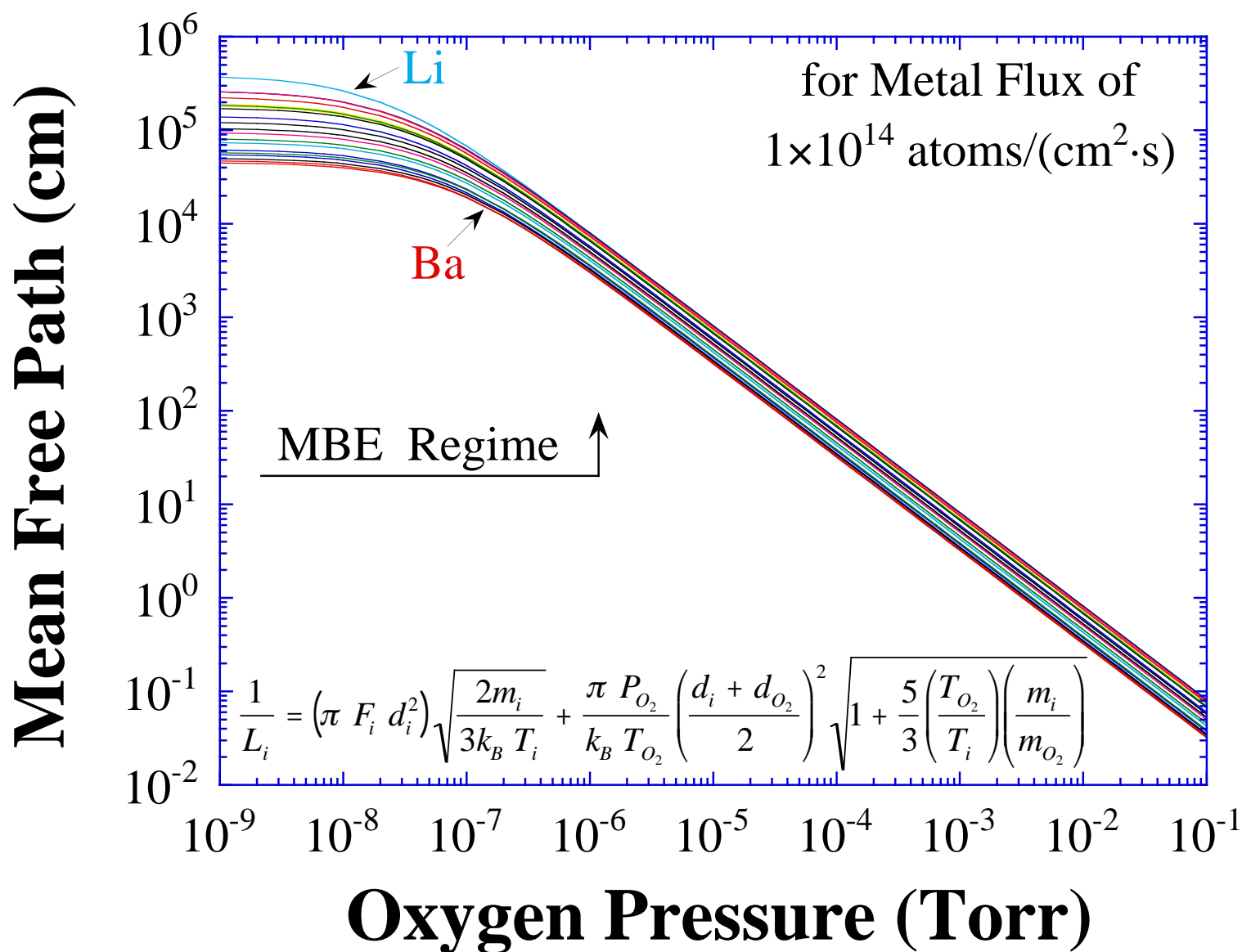
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# Nuts and Bolts of Oxide MBE

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- Mean Free Path (maximum  $P_{O_2}$ )
- Minimum  $P_{O_2}$ , need for  $P_{O_3}$ , Optimal  $T_{\text{sub}}$
- MBE System, Sources, and Crucibles
- Composition Control
  - Adsorption-Controlled Growth
  - Flux-Controlled Growth
- Substrates

# Maximum O<sub>2</sub> Pressure for MBE

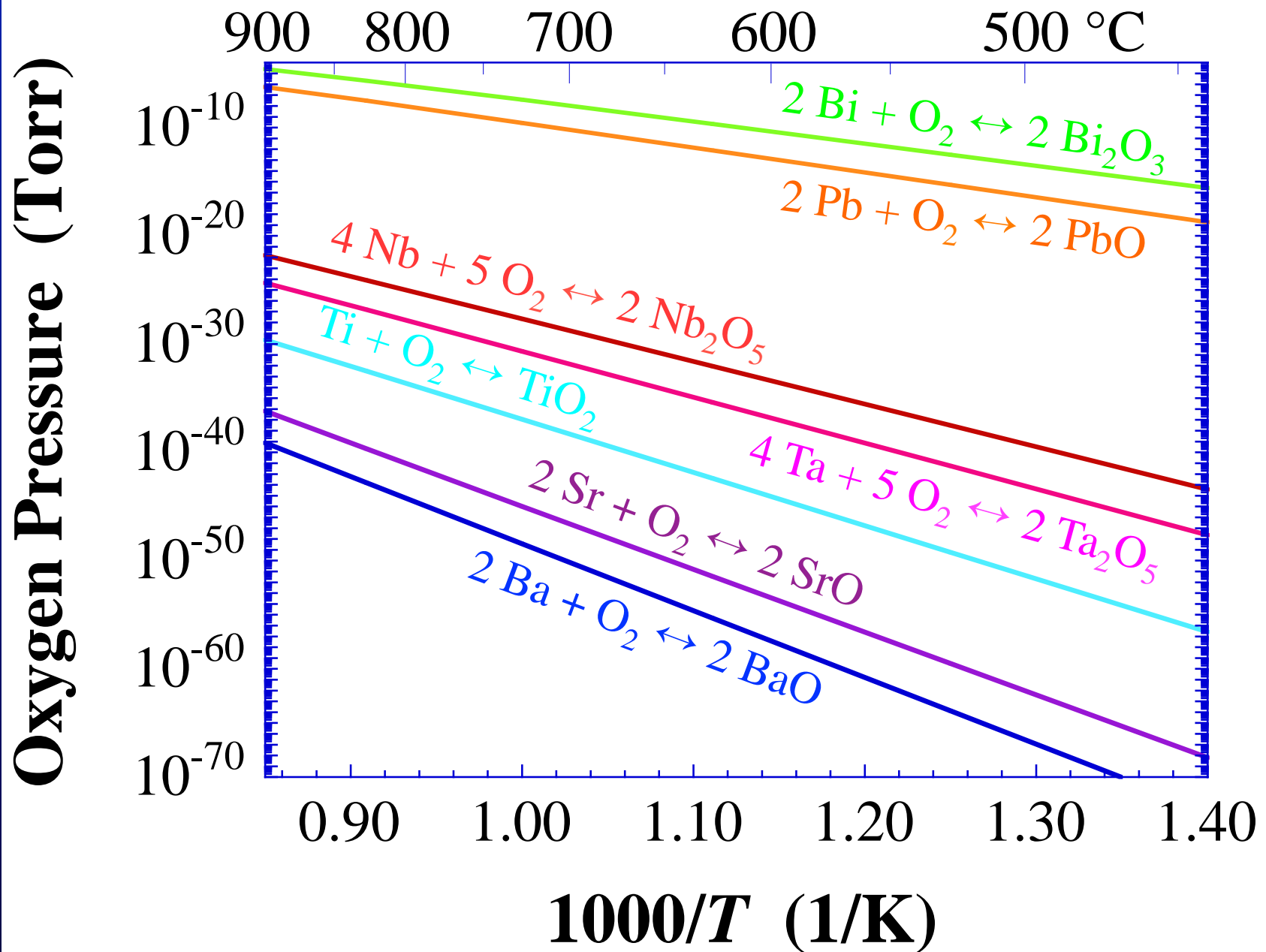


D.G. Schlom and J.S. Harris, Jr., "MBE Growth of High T<sub>c</sub> Superconductors," in:

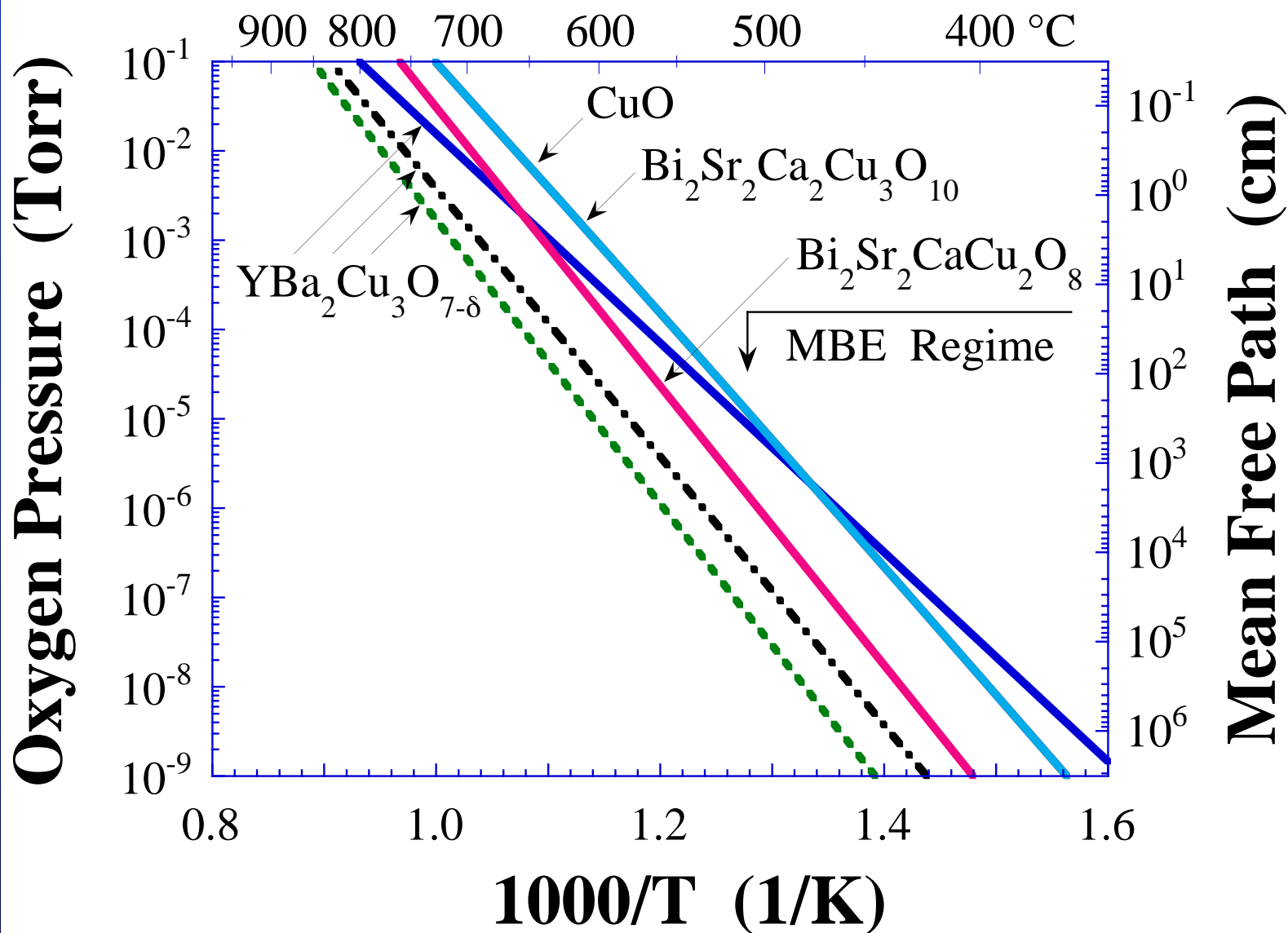
Molecular Beam Epitaxy: Applications to Key Materials, edited by R.F.C. Farrow (Noyes, Park Ridge, 1995), pp. 505-622.



# O<sub>2</sub> Needed to Oxidize Constituents



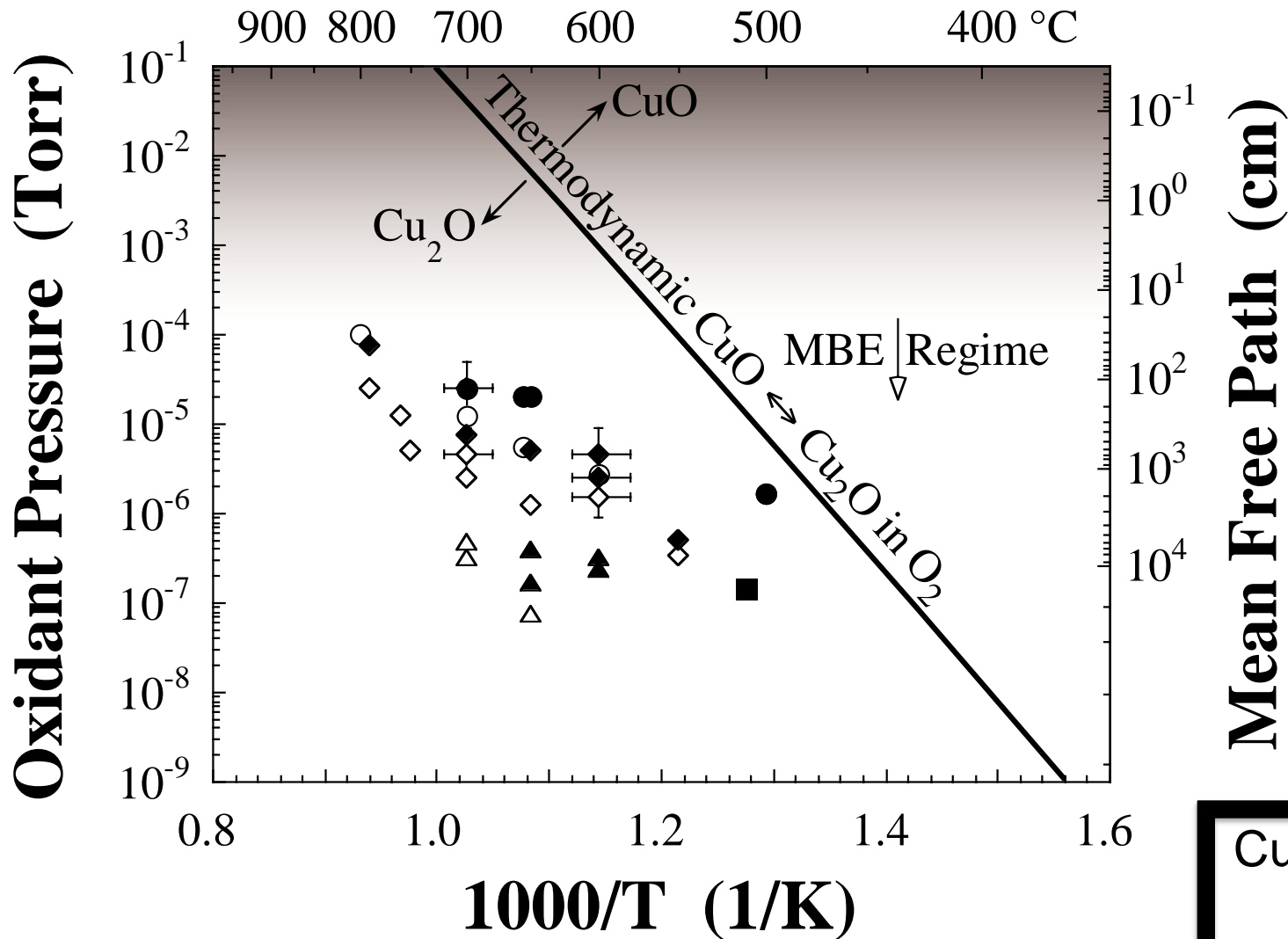
# O<sub>2</sub> Needed to Oxidize Cuprates



D.G. Schlom and J.S. Harris, Jr., "MBE Growth of High  $T_c$  Superconductors," in:

Molecular Beam Epitaxy: Applications to Key Materials, edited by R.F.C. Farrow (Noyes, Park Ridge, 1995), pp. 505-622.

# Superior Oxidants

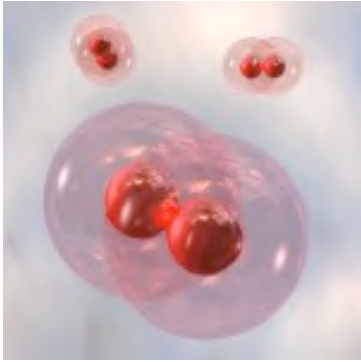


CuO formation in

- ▲ O
- O<sup>+</sup>
- NO<sub>2</sub>
- ◆ O<sub>3</sub>

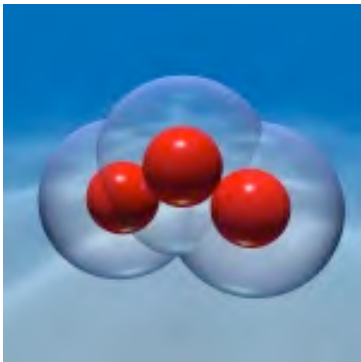
D.G. Schlom and J.S. Harris, Jr., "MBE Growth of High  $T_c$  Superconductors," in: *Molecular Beam Epitaxy: Applications to Key Materials*, edited by R.F.C. Farrow (Noyes, Park Ridge, 1995), pp. 505-622.

# Oxygen vs. Ozone



## Oxygen:

- Easy to use – directly from the cylinder
- Depending on material, films are oxygen-deficient



## Ozone from ozone generator:

- Around 15 wt% O<sub>3</sub> in O<sub>2</sub>
- As O<sub>3</sub> easily decomposes, one O<sub>3</sub> is similar to one O radical
- Higher wt% not achievable, saturation of O<sub>3</sub> concentration

## Distilled ozone:

- Can provide 80-100 wt% pure O<sub>3</sub>
- Better film oxidation, wider process window
- But: Gas is explosive above 10 Torr (absolute), liquid is explosive and shock-sensitive



# Vapor Pressure Oxygen vs. Ozone

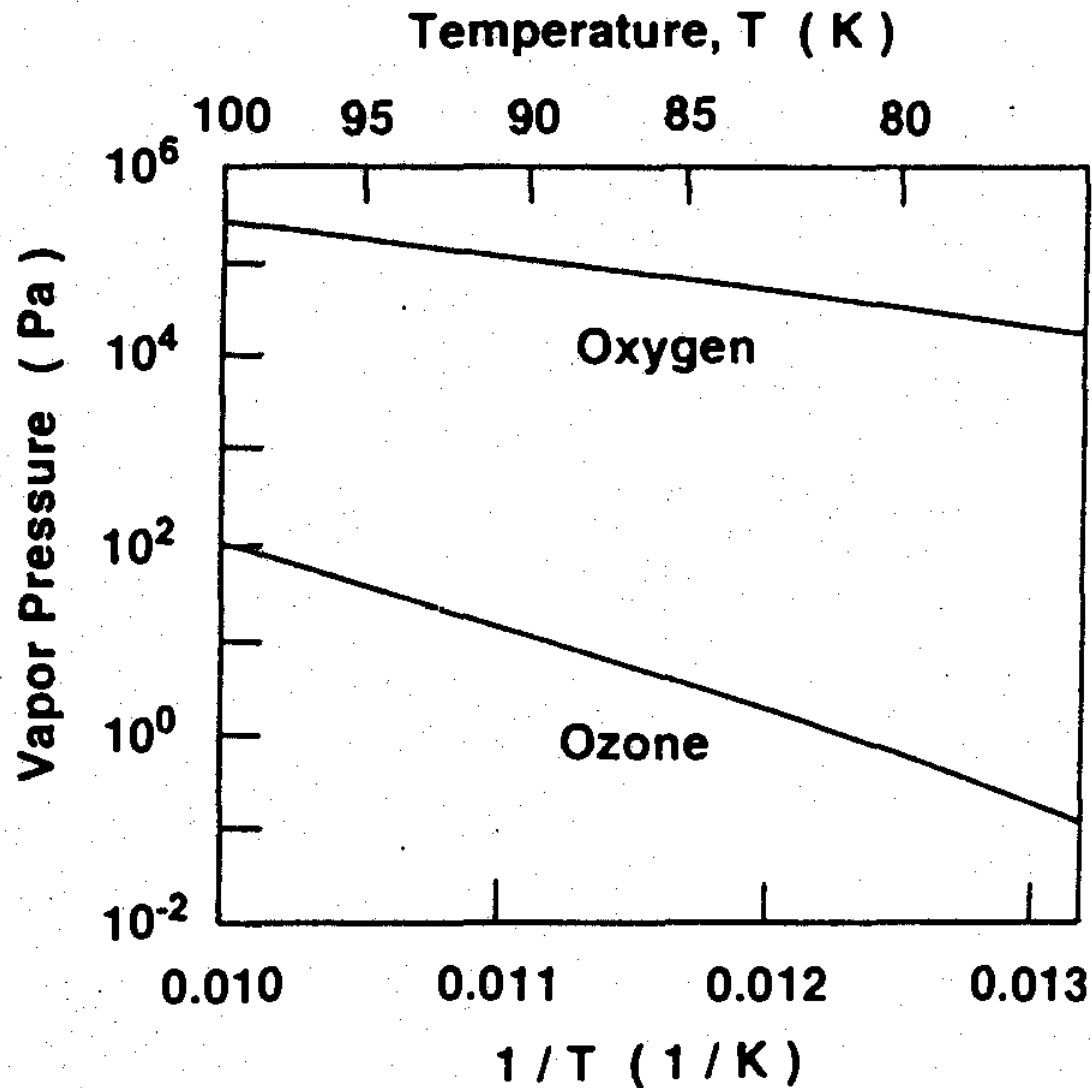


FIG. 4. Vapor pressure of ozone and oxygen as a function of temperature.

S. Hosokawa, and S. Ichimura, "Ozone Jet Generator as an Oxidizing Reagent Source for Preparation of Superconducting Oxide Thin Film," *Review of Scientific Instruments* **62** (1991) 1614-1619.

# Ozone Safety Concerns

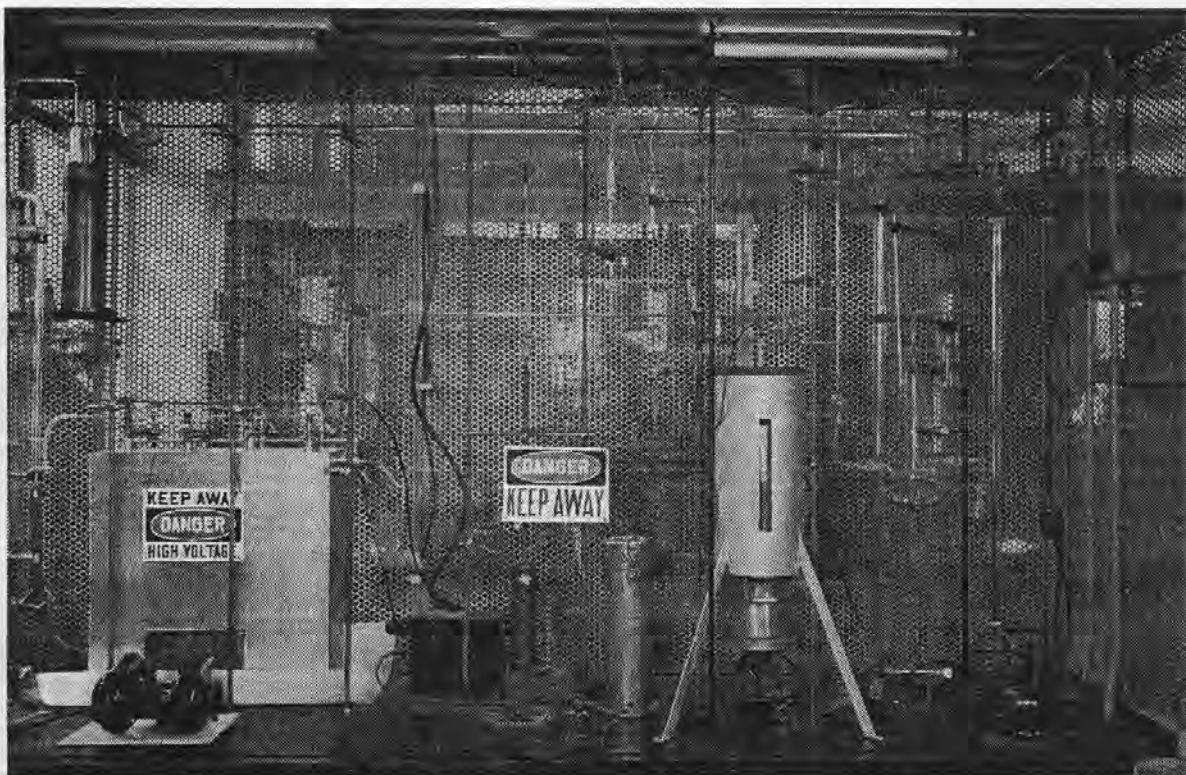


Figure 4. Ozone system

A.C. Jenkins, "Laboratory Techniques for Handling High-Concentration Liquid Ozone," in: *Ozone Chemistry and Technology*, Vol. 21 of *Advances in Chemistry Series*, (American Chemical Society, Washington, D.C., 1959) pp. 13-21.

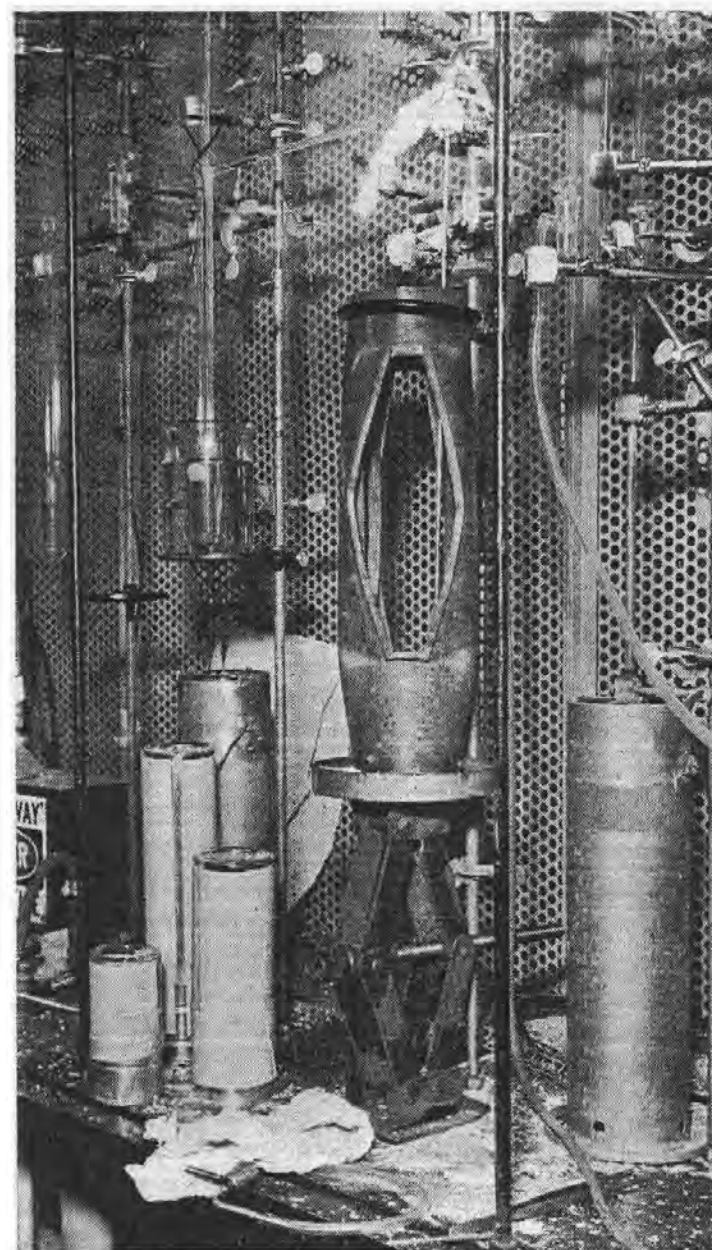


Figure 5. Apparatus after explosion

# Ozone Safety Concerns

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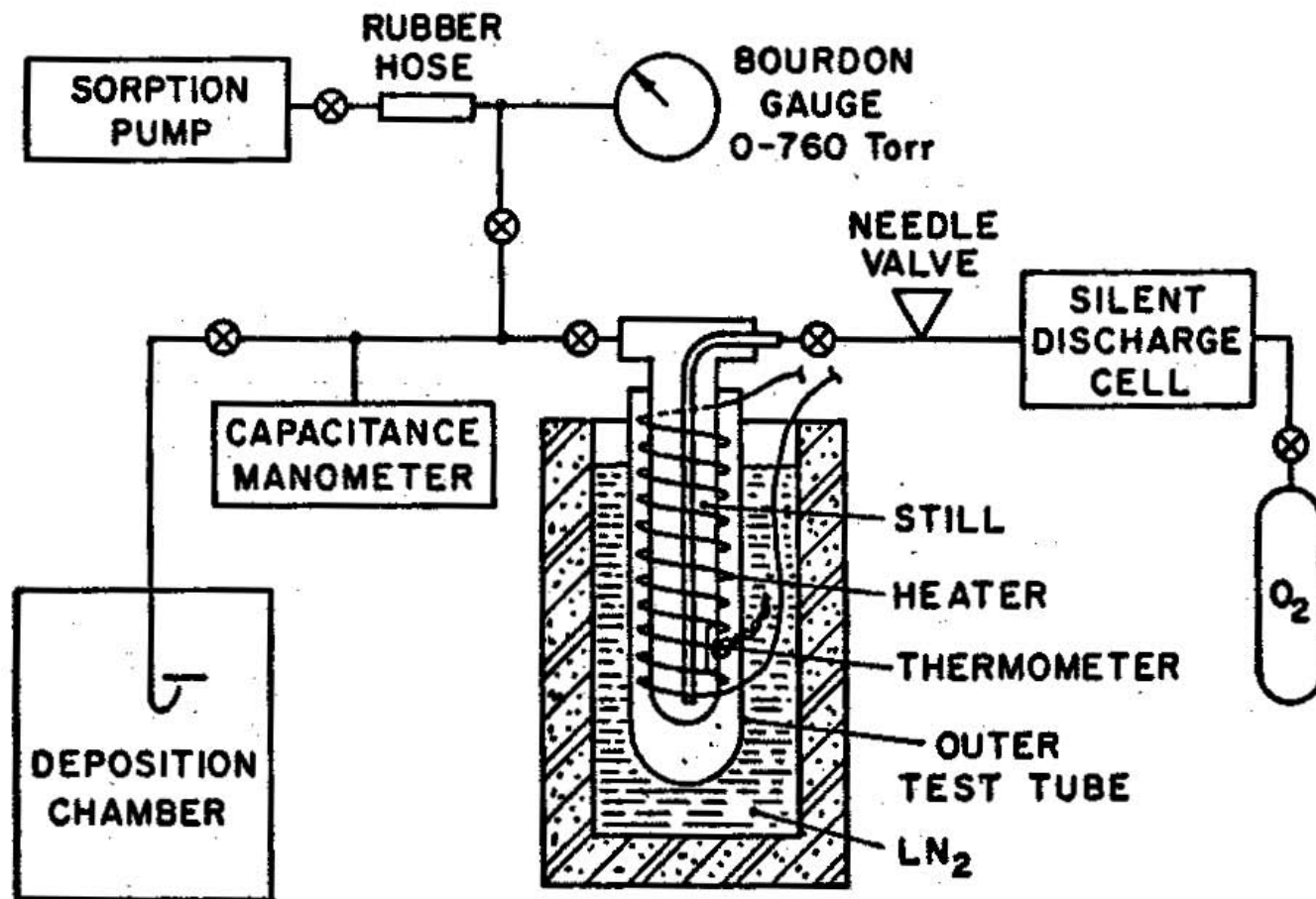
## **OZONE FOR ROCKETS**

Concentrated liquid ozone has been proposed as a rocket fuel by Prof. Clark E. Thorp of the Illinois Institute of Technology, who recently disclosed advances made at the Institute by which ozone can be handled with safety. Ozone is a form of oxygen with three atoms to the molecule instead of two as in ordinary life-supporting oxygen. By demonstrating that it can be safely manufactured, Professor Thorp stated, the door has been opened for tonnage production. During World War II, German scientists worked overtime on an ozone-propelled rocket designed to bombard New York City from European launching platforms. But they were unable to discover the secret of handling ozone without spontaneous detonation.

“Ozone for Rockets,”

*Ordnance: The Journal of the Army Ordnance Association* **36** [187]  
(July-August 1951) pp. 108-110.

# Ozone Distillation (U. Minnesota)



**FIG. 1. Ozone-production and storage apparatus. During production of ozone, the still heater is off and the outer test tube surrounding the still is filled with liquid nitrogen.**

D.D. Berkley, A.M. Goldman, B.R. Johnson, J. Morton, and T. Wang,  
“Techniques for the Growth of Superconducting Oxide Thin Films Using Pure Ozone Vapor,”  
*Review of Scientific Instruments* **60** (1989) 3769-3774.

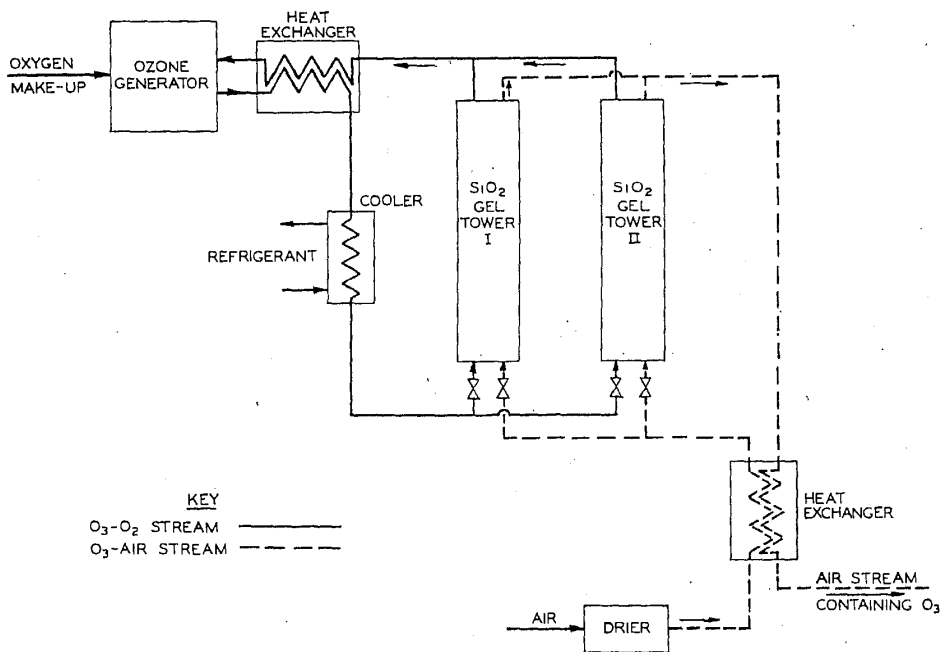


# Ozone in Silica Gel–Union Carbide

## Separation of Ozone from Oxygen by a Sorption Process

G. A. COOK, A. D. KIFFER, C. V. KLUMPP, A. H. MALIK, and L. A. SPENCE

Research and Development Laboratory, Linde Co., A Division of Union Carbide Corp.,  
Tonawanda, N. Y.



Ozone is separated from oxygen by adsorption on refrigerated silica gel, followed by desorption, either in pure form at reduced pressure, or diluted by air, nitrogen, argon, or other gas not strongly adsorbed on silica gel. This is a practical method, free from hazard when correctly performed.

Figure 7. Simplified flow diagram for two-stage transfer process

G.A. Cook, A.D. Kiffer, C.V. Klumpp, A.H. Malik, and L.A. Spence,

“Separation of Ozone from Oxygen by a Sorption Process,” in: *Ozone Chemistry and Technology*, Vol. 21 of Advances in Chemistry Series, (American Chemical Society, Washington, D.C., 1959) pp. 44-52.

# Ozone in Silica Gel—Union Carbide

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## Safety Tests

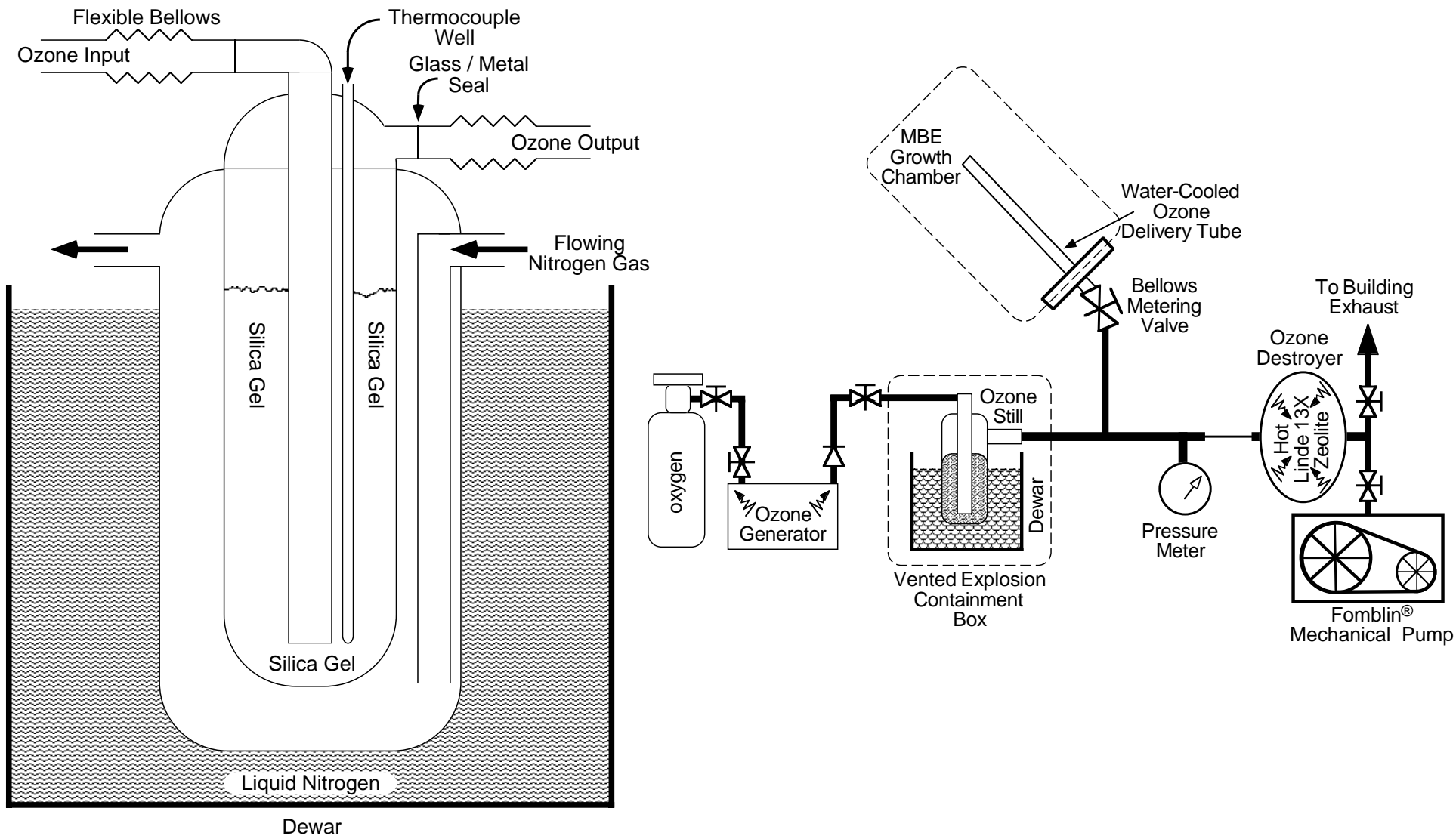
An electric spark was passed between two tungsten electrodes within the gel bed. No explosions took place at  $-105^{\circ}\text{C}$ . with loadings as high as 21 grams of ozone per 100 grams of gel.

A piece of fine iron wire was placed in the bed of gel and ignited by passing an electric current through it. With a loading of 9 grams of ozone per 100 grams of gel, a test was made at  $-78^{\circ}\text{C}$ . After ignition of the wire, a glow in the vessel was observed, but no explosion took place. With a loading of 20 grams of ozone per 100 grams of gel at  $-105^{\circ}\text{C}$ ., a dull pop was heard when the wire was ignited, most of the ozone changed to oxygen, and the gel was extensively pulverized. The glass container was broken but not shattered.

G.A. Cook, A.D. Kiffer, , C.V. Klumpp, A.H. Malik, and L.A. Spence,

“Separation of Ozone from Oxygen by a Sorption Process,” in: *Ozone Chemistry and Technology*, Vol. 21 of Advances in Chemistry Series, (American Chemical Society, Washington, D.C., 1959) pp. 44-52.

# Ozone in Silica Gel



D.G. Schlom and J.S. Harris, Jr., "MBE Growth of High  $T_c$  Superconductors," in:

*Molecular Beam Epitaxy: Applications to Key Materials*, edited by R.F.C. Farrow (Noyes, Park Ridge, 1995), pp. 505-622.

# Process Control



- **Safety committee requirements led to development of fully integrated process controller**
- **Controller monitors all process equipment and parameters**
- **Countermeasures if close to critical limit**
  - **Self-test of equipment before process start**
    - **Fully automated operation**
- **No need to manually watch ozone process, user can focus on MBE growth!!!**

# Ozone System Passivation

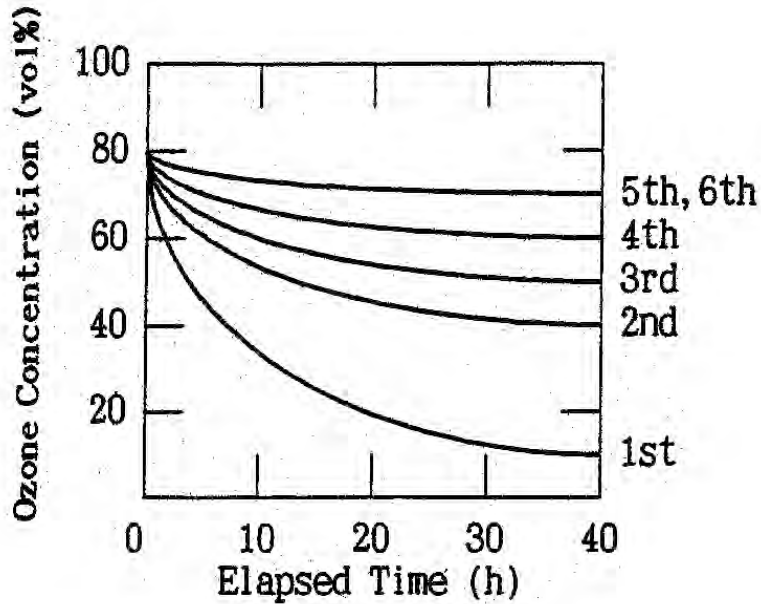


Fig. 2. Ozone concentration decay during ozone passivation.

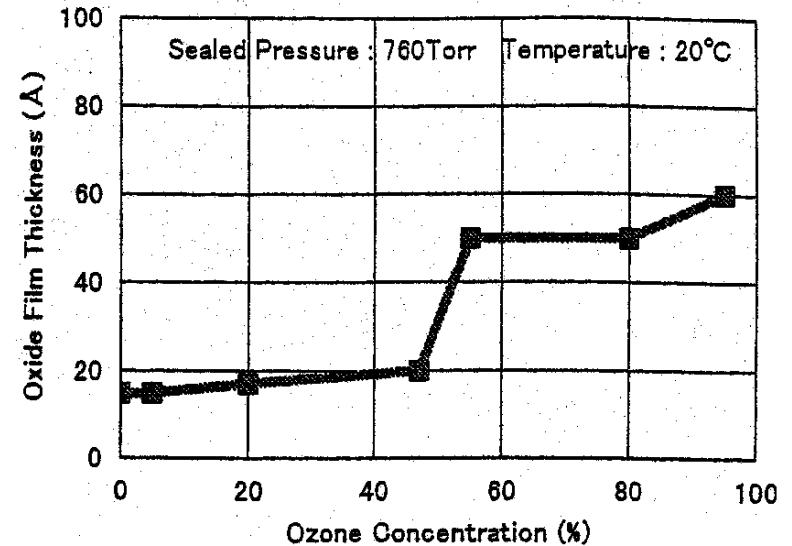


Fig. 4. Relationship of ozone concentration with thickness of passivated film formed at atmospheric pressure and 20°C.

K. Koike, G. Inoue, T. Takata, and T. Fukuda,  
Ozone Passivation Technique for Corrosive Gas Distribution System,”  
*Japanese Journal of Applied Physics* 36 (1997) 7437-7441.

# Pros and Cons of Ozone

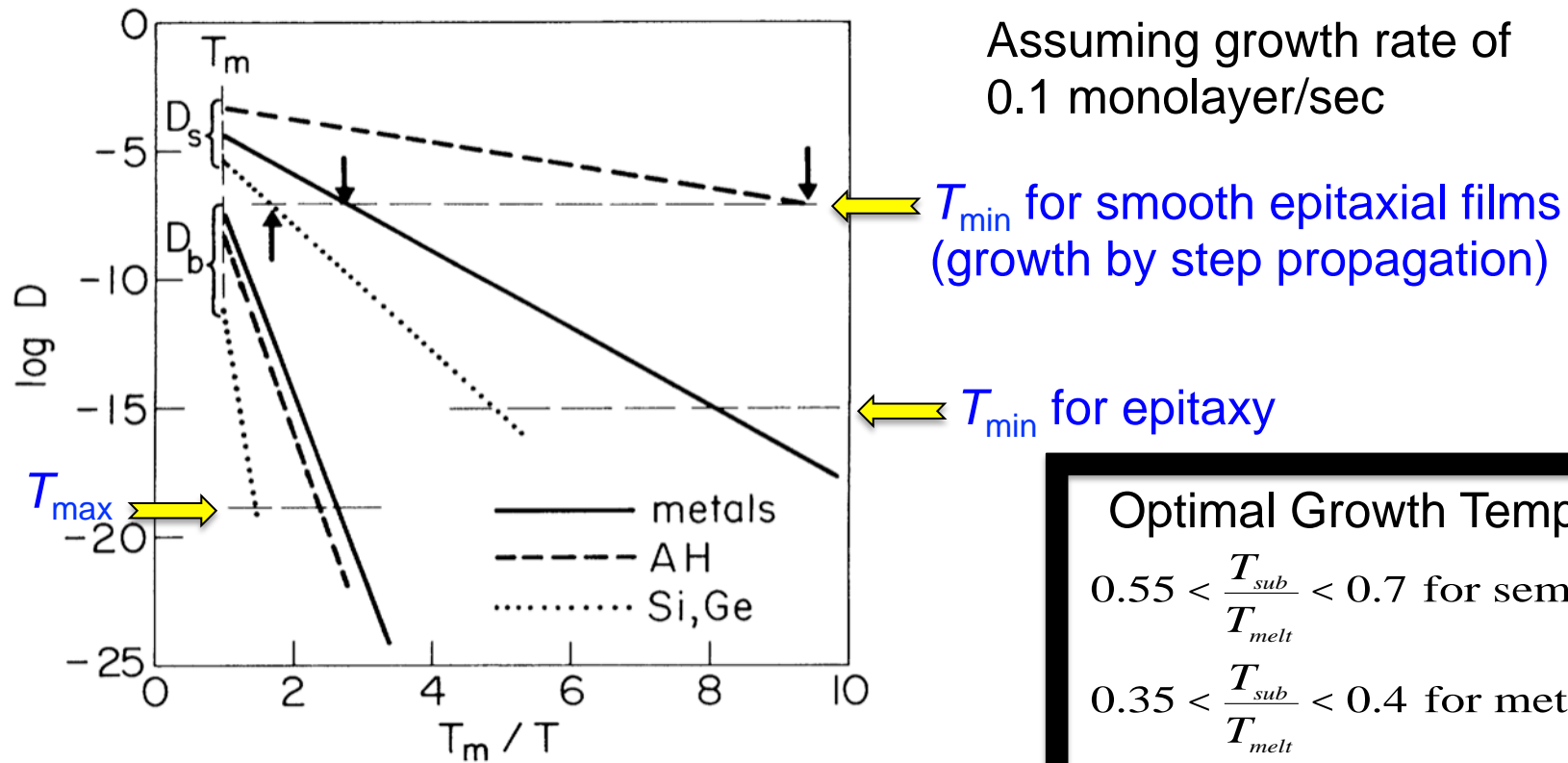
- Pros
  - Excellent Oxidant  
(about 1000× more powerful than O<sub>2</sub>)
  - 80% Ozone (+20% O<sub>2</sub>) Delivery Possible to the Substrate
  - No Energetic Species (thermal ozone beam)
  - Inexpensive (if you make it yourself)
- Cons
  - Safety (Ozone still issues)
  - Safety (Pump issues)
  - Need Ozone-Compatible UHV Leak Valve
  - Need to Passivate Ozone System

# Nuts and Bolts of Oxide MBE

---

- Mean Free Path (maximum  $P_{O_2}$ )
- Minimum  $P_{O_2}$ , need for  $P_{O_3}$ , Optimal  $T_{\text{sub}}$
- MBE System, Sources, and Crucibles
- Composition Control
  - Adsorption-Controlled Growth
  - Flux-Controlled Growth
- Substrates

# Surface vs. Bulk Diffusion



## Optimal Growth Temperatures

$$0.55 < \frac{T_{sub}}{T_{melt}} < 0.7 \text{ for semiconductors}$$

$$0.35 < \frac{T_{sub}}{T_{melt}} < 0.4 \text{ for metals}$$

$$0.1 < \frac{T_{sub}}{T_{melt}} < 0.4 \text{ for simple ceramics}$$

FIG. 6. Diagram showing deduced global dependences of surface and bulk diffusion coefficients,  $D_s$  and  $D_b$ , on  $T_m/T$  for metals (solid lines), elemental semiconductors (dotted lines), and salts (dashed lines). The construction is described in the text. Smooth flat interfaces generally require  $D_s \gtrsim 10^{-8} - 10^{-7} \text{ cm}^2/\text{sec}$ , which fixes the lowest growth temperatures (arrows) as  $\sim 3T_m/8$ ,  $0.55T_m$ , and  $0.1T_m$  in the three cases. RHEED oscillations are expected for  $D_s \gtrsim 10^{-15} \text{ cm}^2/\text{sec}$  and bulk interdiffusion for  $D_b \gtrsim 10^{-19} \text{ cm}^2/\text{sec}$ .

M.H. Yang and C.P. Flynn,  
*Physical Review Letters* **62**  
(1989) 2476-2479.



# Universal Diffusion Behavior of Metals

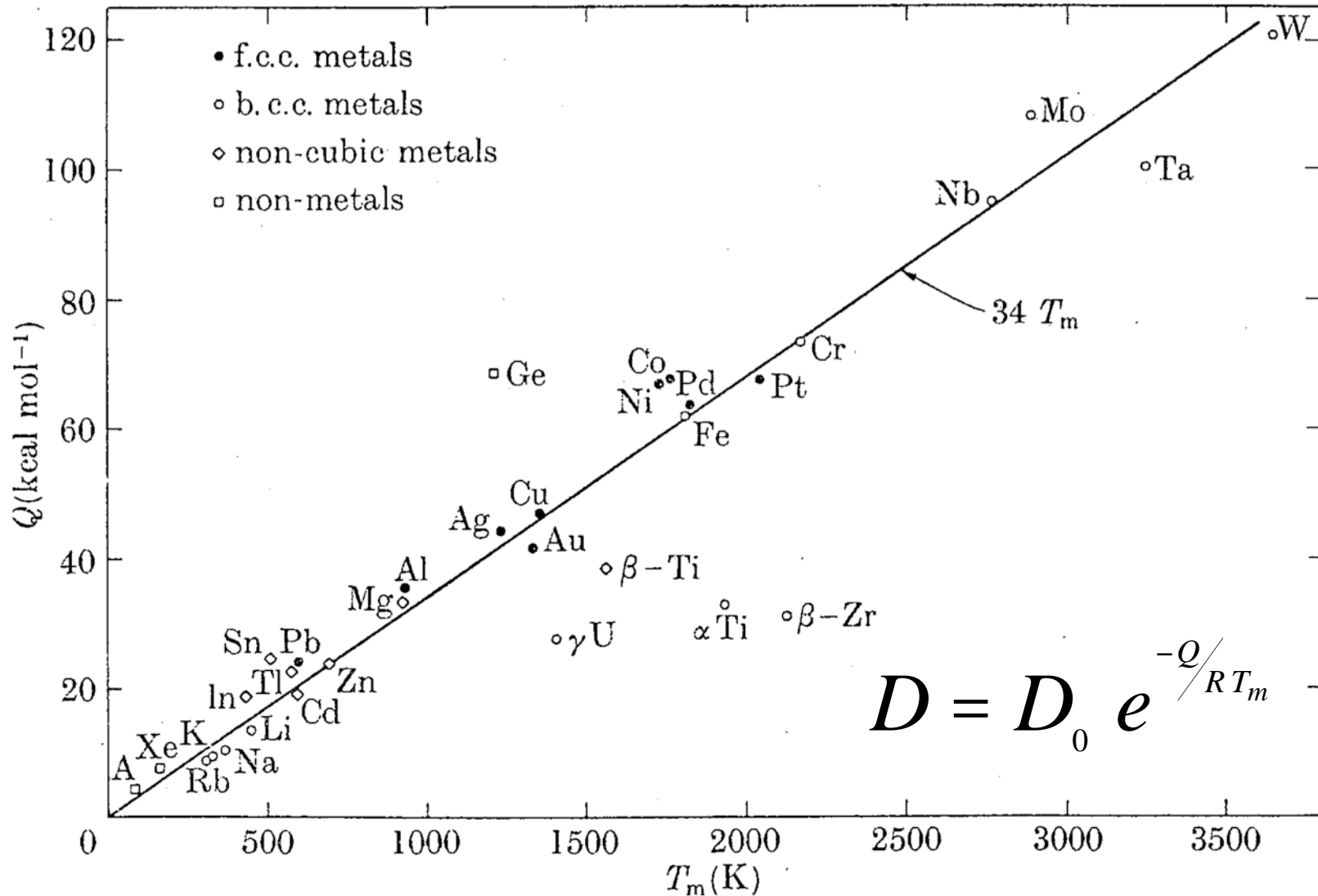
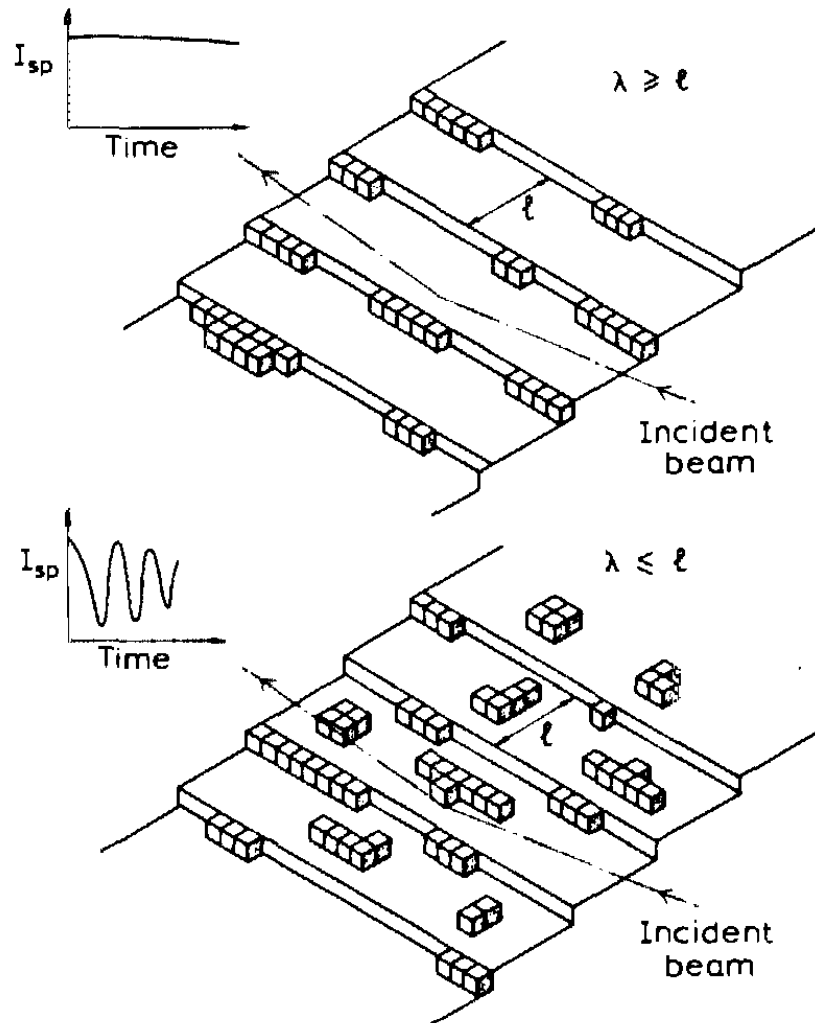


FIG. 14.31. The correlation between  $Q$  and  $T_m$ .

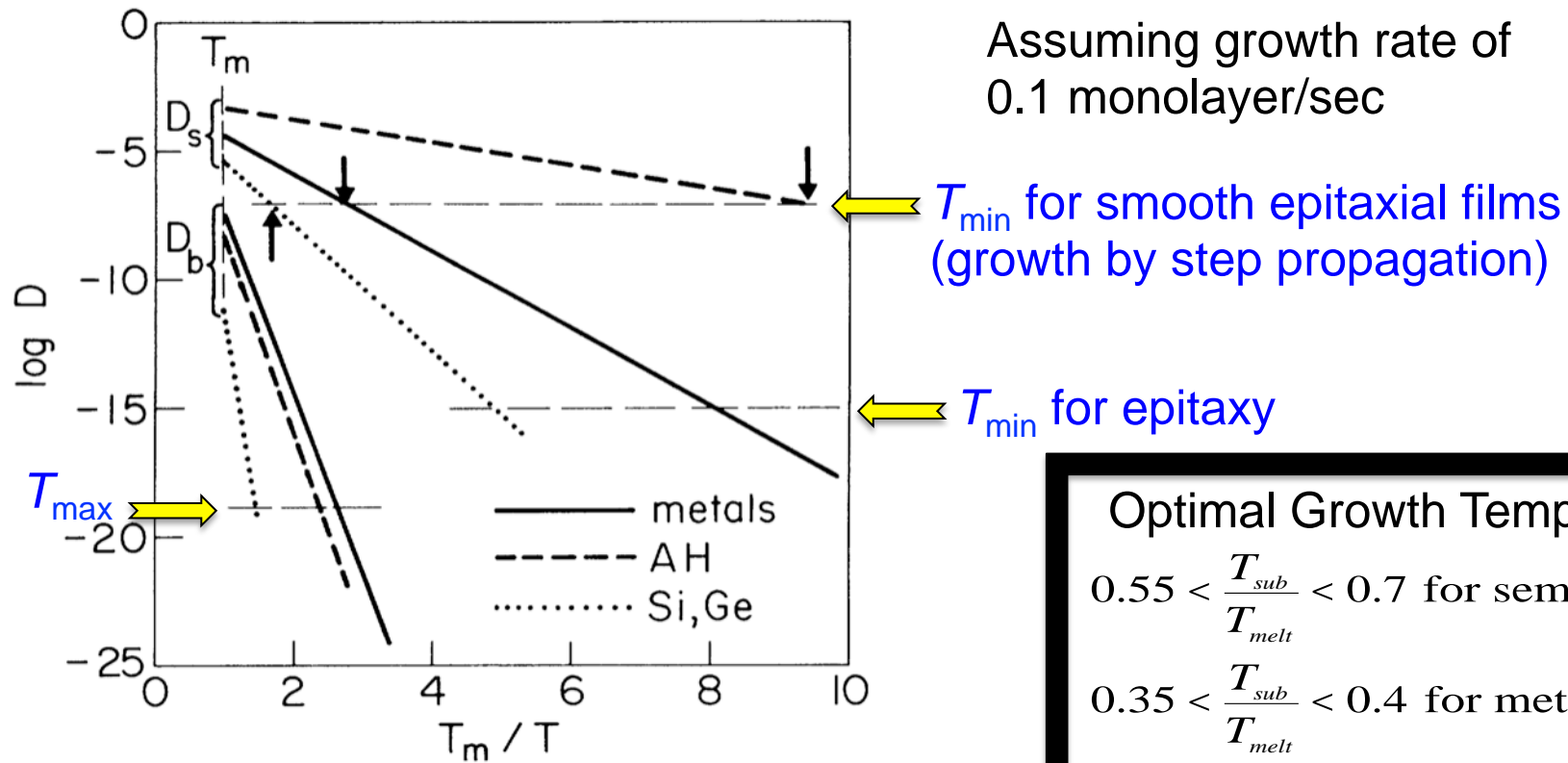
# Determining Surface Diffusion from RHEED Oscillations



J.H. Neave, P.J. Dobson, B.A. Joyce, and J. Zhang,  
*Applied Physics Letters* **47** (1985) 100-102.

FIG. 1. Schematic illustration of the principle of the method, showing the change in RHEED information as the growth mode changes from "step flow" to 2-D nucleation. Steps lie along [100].

# Surface vs. Bulk Diffusion



## Optimal Growth Temperatures

$$0.55 < \frac{T_{sub}}{T_{melt}} < 0.7 \text{ for semiconductors}$$

$$0.35 < \frac{T_{sub}}{T_{melt}} < 0.4 \text{ for metals}$$

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FIG. 6. Diagram showing deduced global dependences of surface and bulk diffusion coefficients,  $D_s$  and  $D_b$ , on  $T_m/T$  for metals (solid lines), elemental semiconductors (dotted lines), and salts (dashed lines). The construction is described in the text. Smooth flat interfaces generally require  $D_s \gtrsim 10^{-8} - 10^{-7} \text{ cm}^2/\text{sec}$ , which fixes the lowest growth temperatures (arrows) as  $\sim 3T_m/8$ ,  $0.55T_m$ , and  $0.1T_m$  in the three cases. RHEED oscillations are expected for  $D_s \gtrsim 10^{-15} \text{ cm}^2/\text{sec}$  and bulk interdiffusion for  $D_b \gtrsim 10^{-19} \text{ cm}^2/\text{sec}$ .

M.H. Yang and C.P. Flynn,  
*Physical Review Letters* **62**  
(1989) 2476-2479.

# Surface Energy Considerations

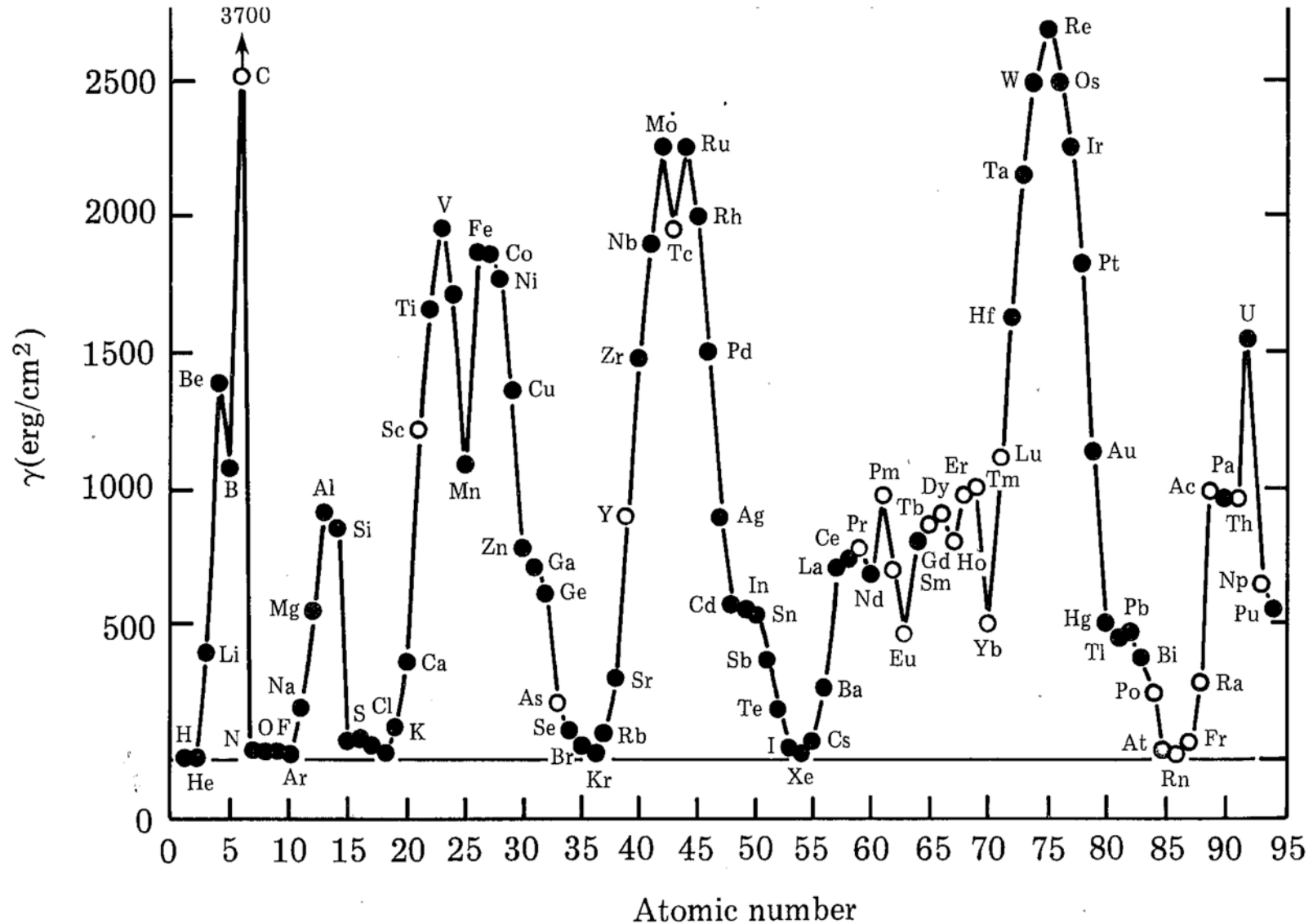
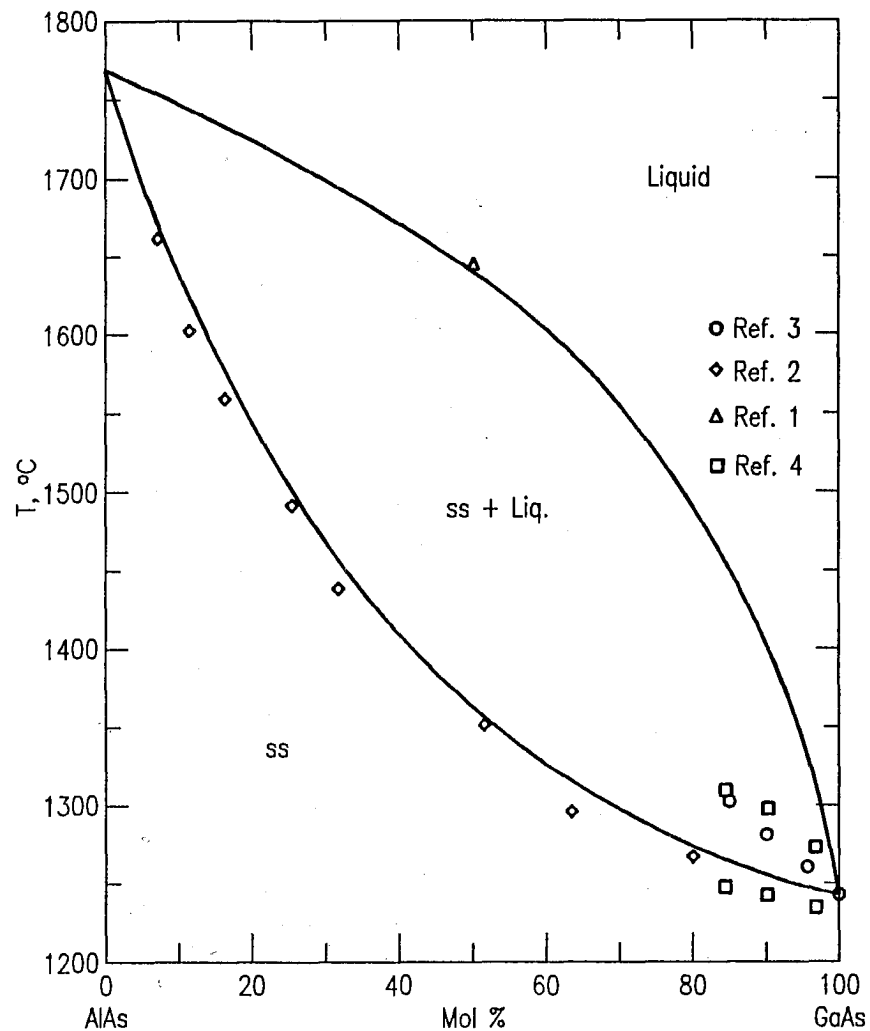


FIGURE 2.7 Surface tension of elements in the liquid phase (from Zangwill, 1988).

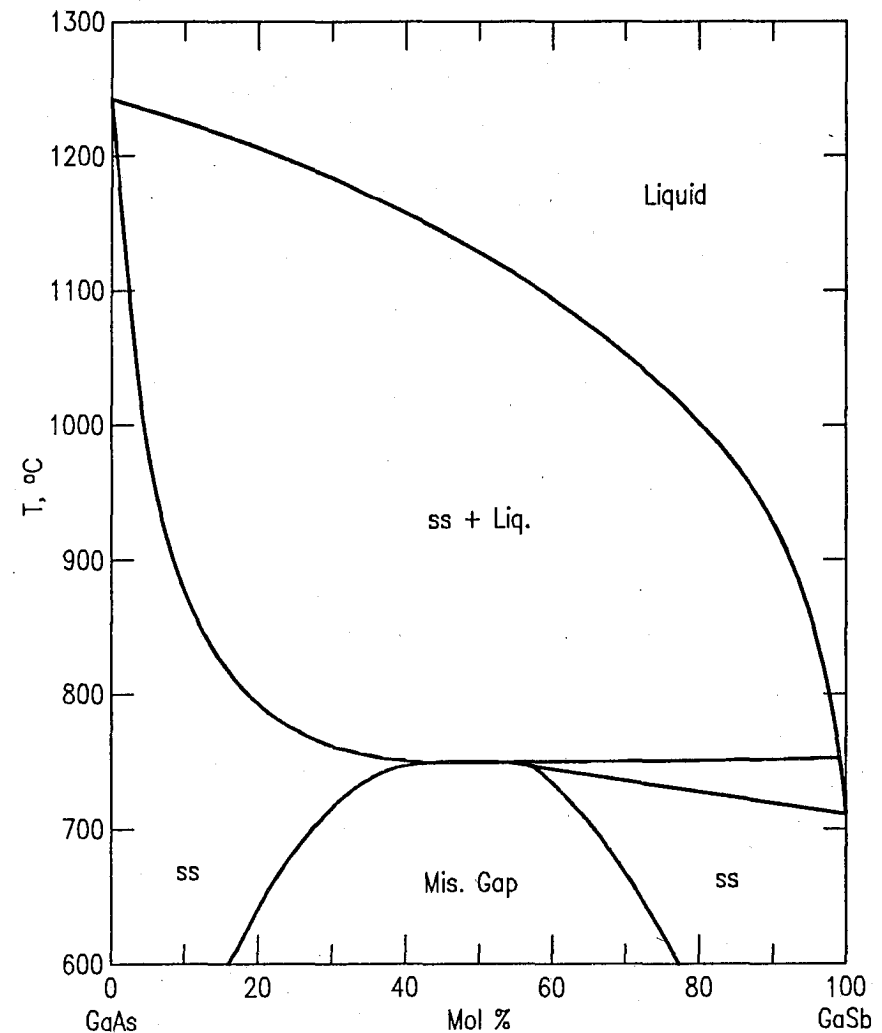
K.-N. Tu, J.W. Mayer, and L.C. Feldman,

*Electronic Thin Film Science for Electrical Engineers and Materials Scientists* (Macmillan, 1992).

# Thermodynamic Considerations



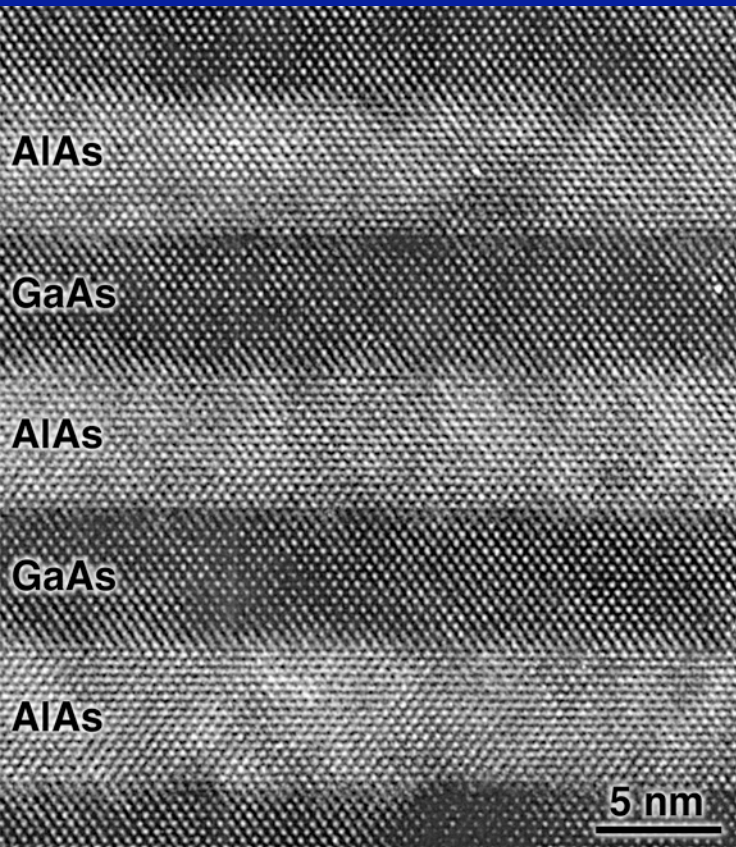
**Fig. 8343—Pseudobinary system AlAs-GaAs.**  
K. Y. Ma, S. H. Li, and G. B. Stringfellow, "P, As, and Sb Phase Diagrams", Special Report to the Standard Reference Data Program, National Institute of Standards and Technology; Gaithersburg, Maryland (1987).



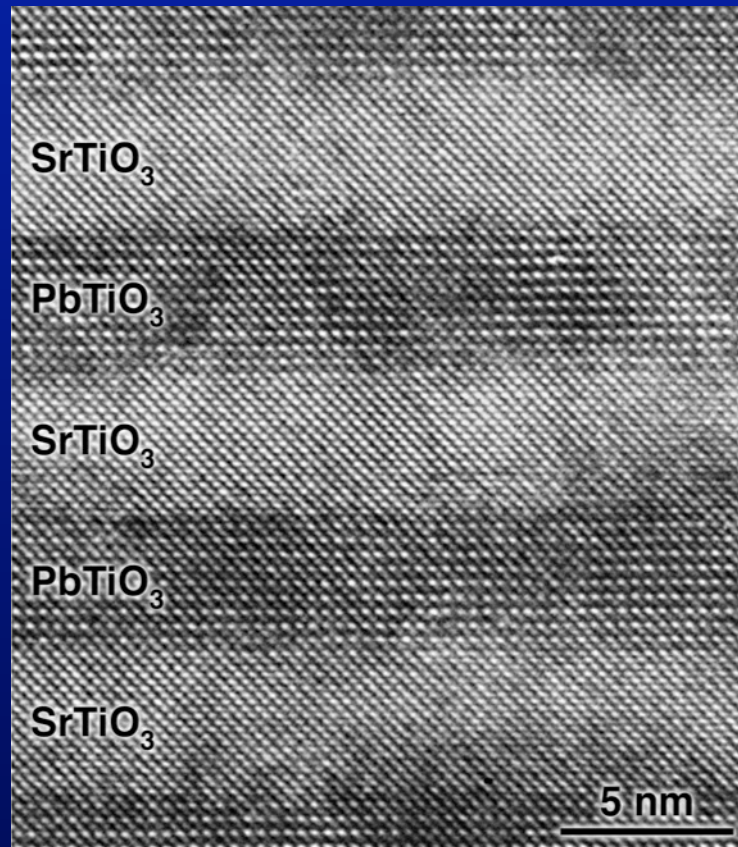
**Fig. 8362—Pseudobinary system GaAs-GaSb. Calculated diagram.**  
K. Y. Ma, S. H. Li, and G. B. Stringfellow, "P, As, and Sb Phase Diagrams", Special Report to the Standard Reference Data Program, National Institute of Standards and Technology; Gaithersburg, Maryland (1987).

# TEM of MBE-Grown Superlattices

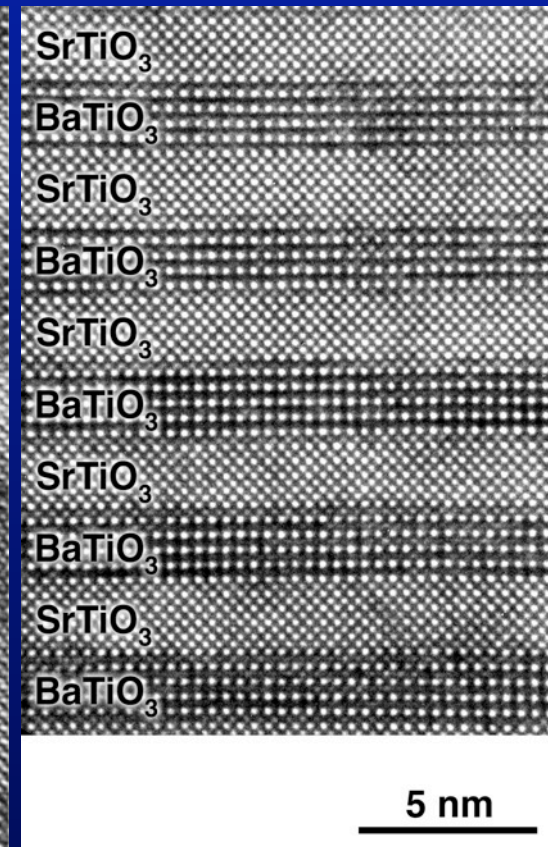
AIAs / GaAs



PbTiO<sub>3</sub> / SrTiO<sub>3</sub>



BaTiO<sub>3</sub> / SrTiO<sub>3</sub>



A.K. Gutakovskii *et al.*,  
Phys. Stat. Sol. (a) **150** (1995) 127.

C.D. Theis  
(1<sup>st</sup> Generation Schlom Group)  
HRTEM—Pan Group (Michigan)

J.H. Haeni  
(2<sup>nd</sup> Generation)

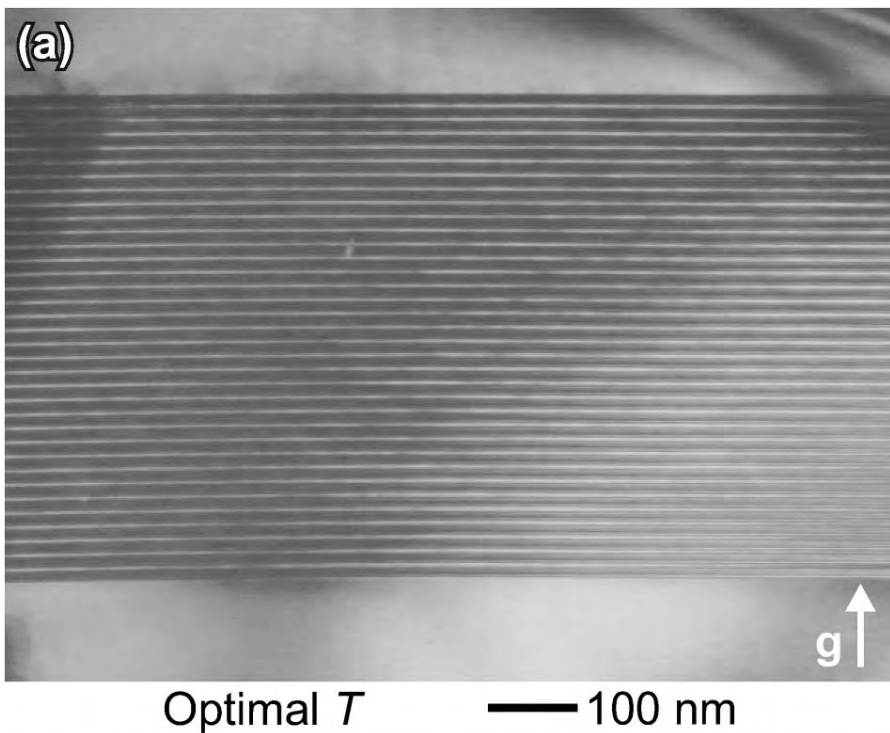
D.G. Schlom *et al.*, Mater. Sci. Eng. B **87** (2001) 282.

From the observed  
morphology,  
which likely has the  
higher surface energy?

(a) GaAs

(b) AlAs

(c) They appear to have the same  
surface energies



# Increased Interface Roughness and Clustering at Non-Optimal Growth Conditions

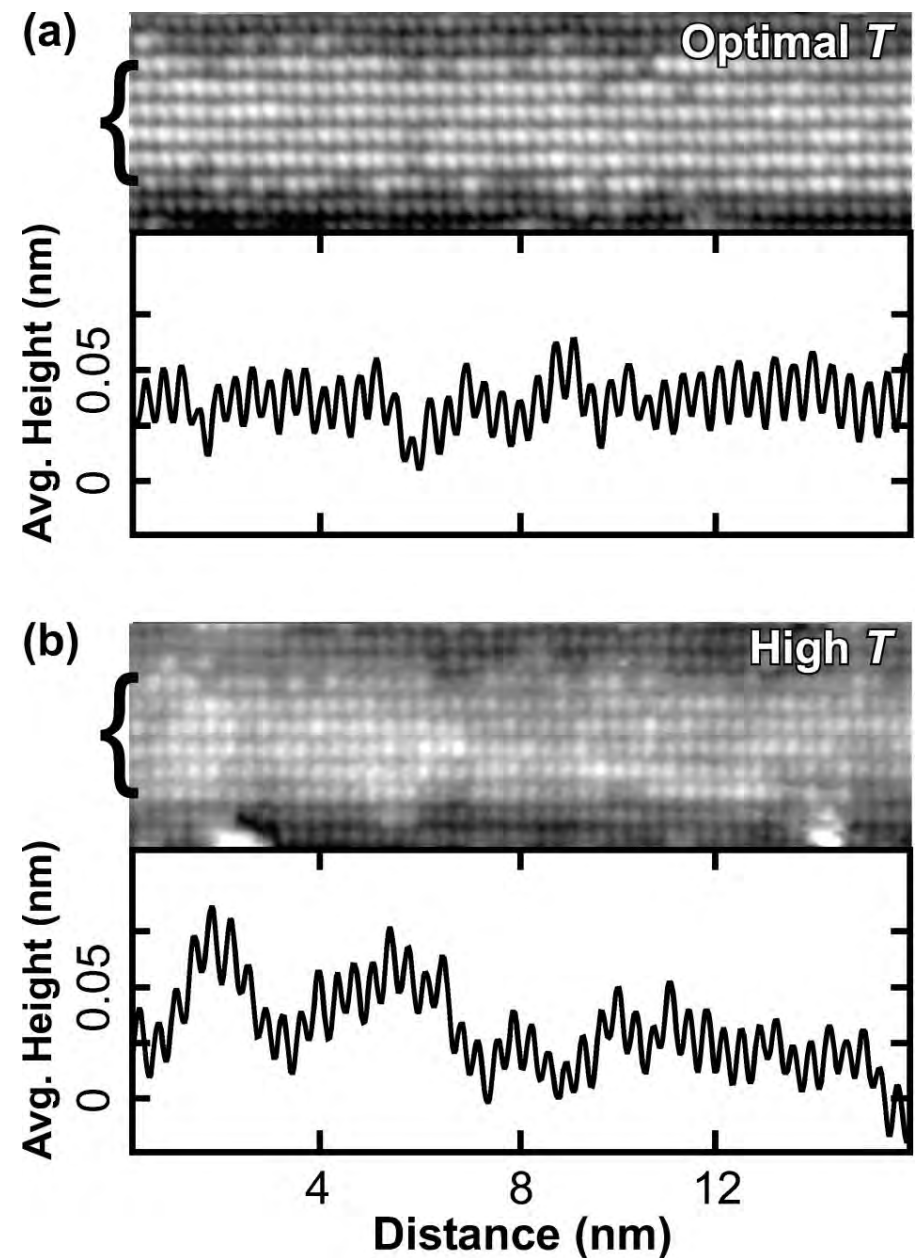


FIG. 6. XSTM images and average-height profiles for InGaSb-alloy layers in an (a) optimal- and (b) high-temperature sample. (a)  $-2.0$  V, 50 pA and (b)  $-2.5$  V, 0.5 nA.

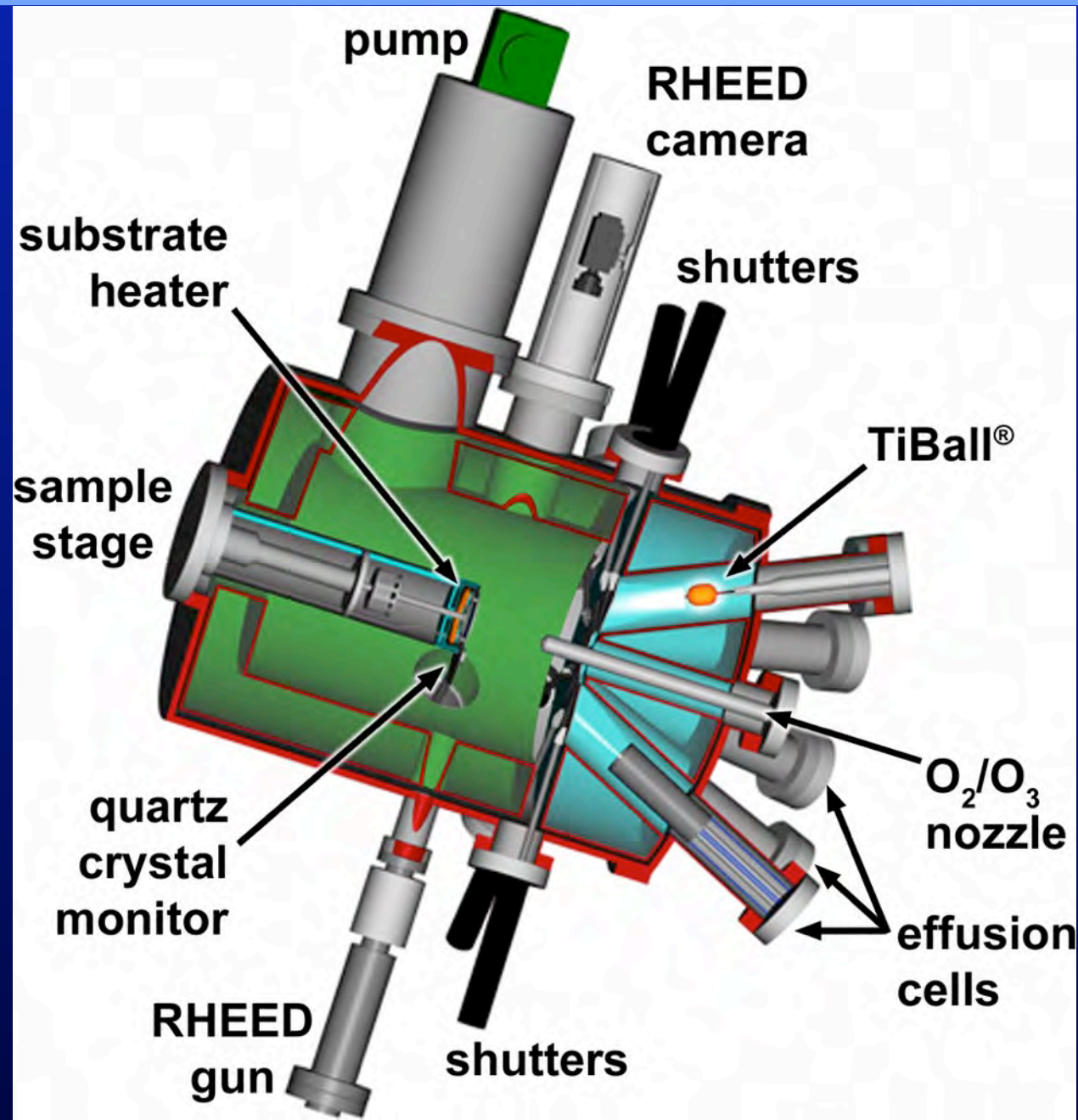


# Nuts and Bolts of Oxide MBE

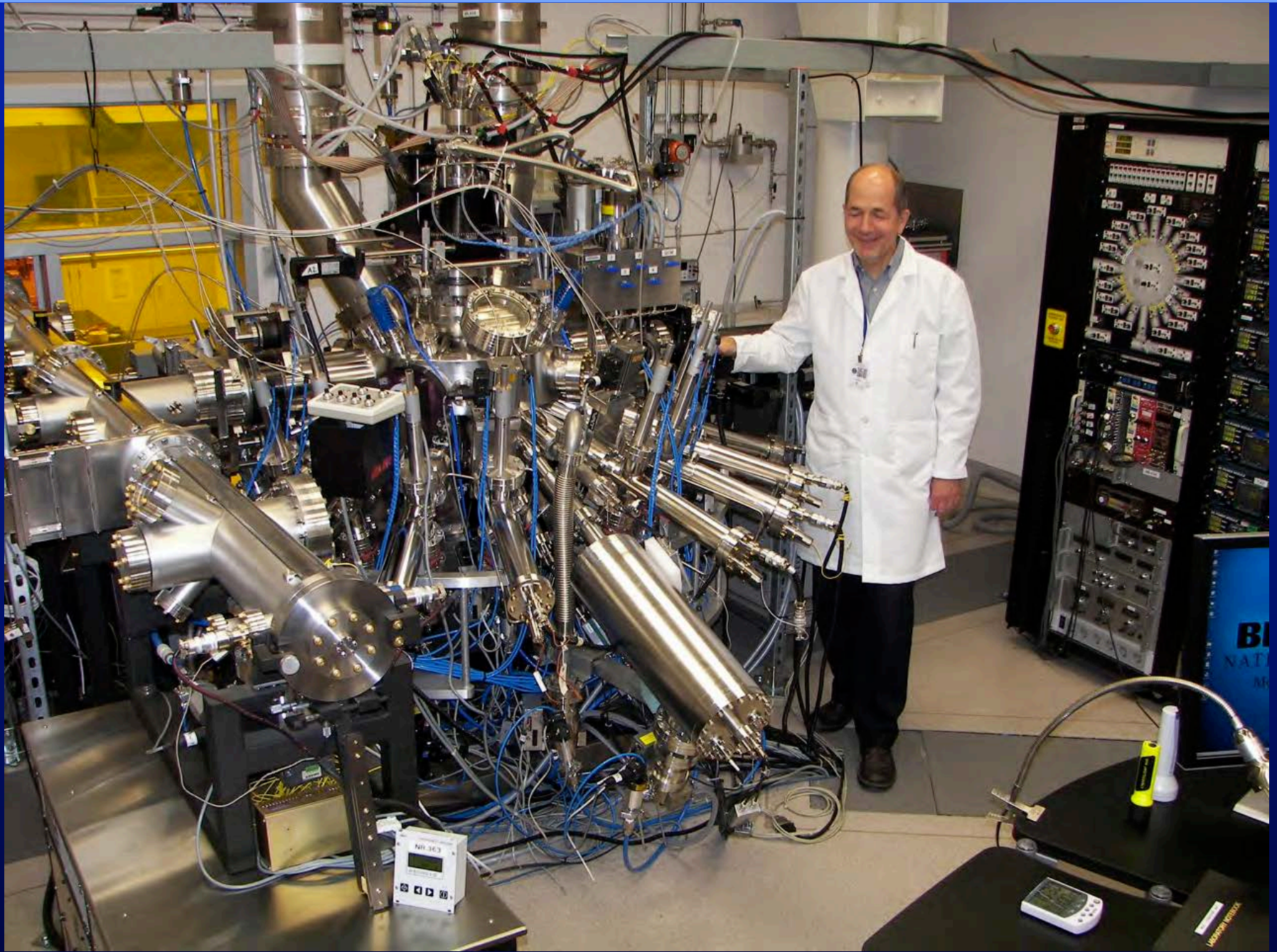
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- Mean Free Path (maximum  $P_{O_2}$ )
- Minimum  $P_{O_2}$ , need for  $P_{O_3}$ , Optimal  $T_{sub}$
- MBE System, Sources, and Crucibles
- Composition Control
  - Adsorption-Controlled Growth
  - Flux-Controlled Growth
- Substrates

# Oxide MBE System

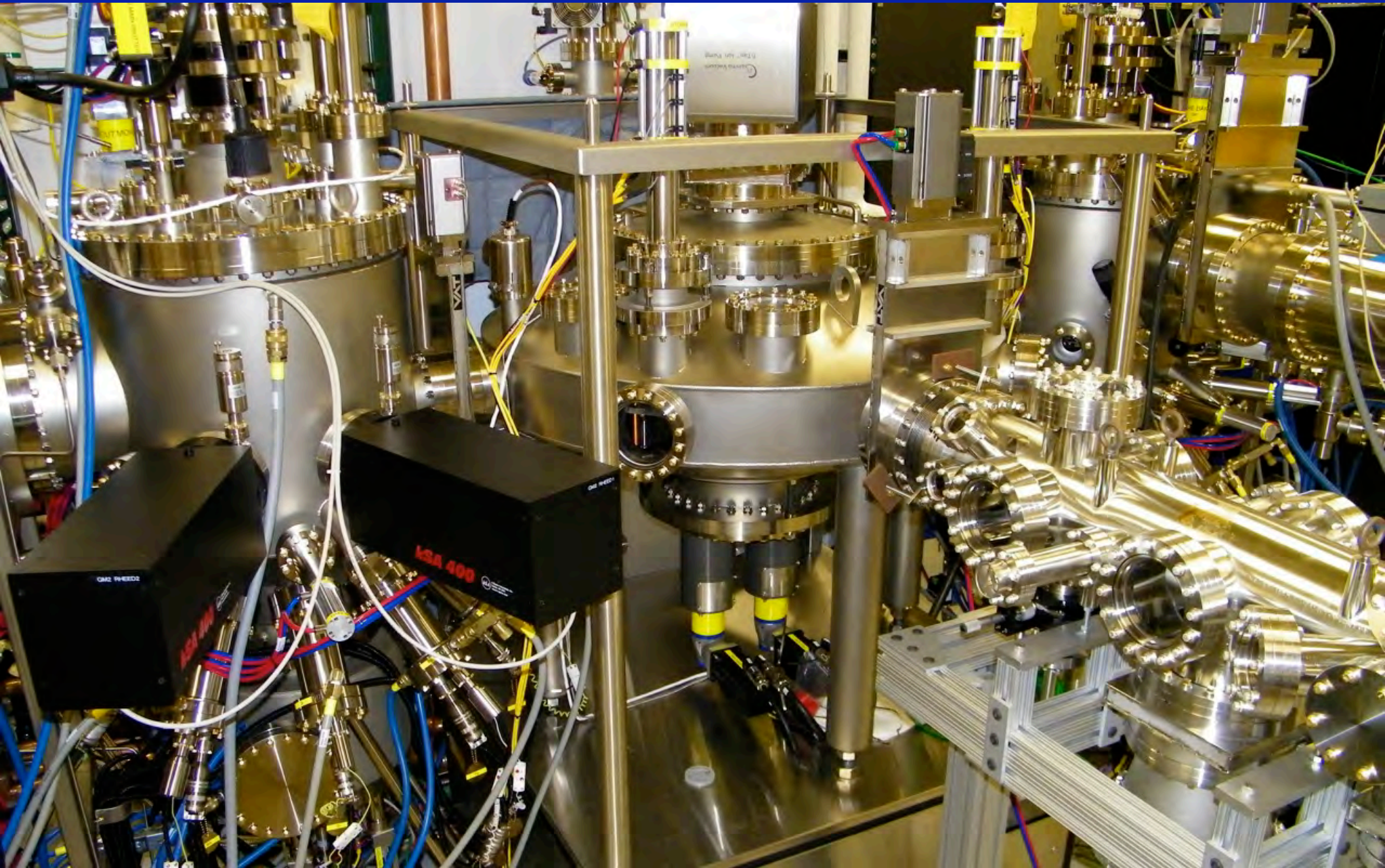


# Oxide MBE at Brookhaven Nat. Lab.

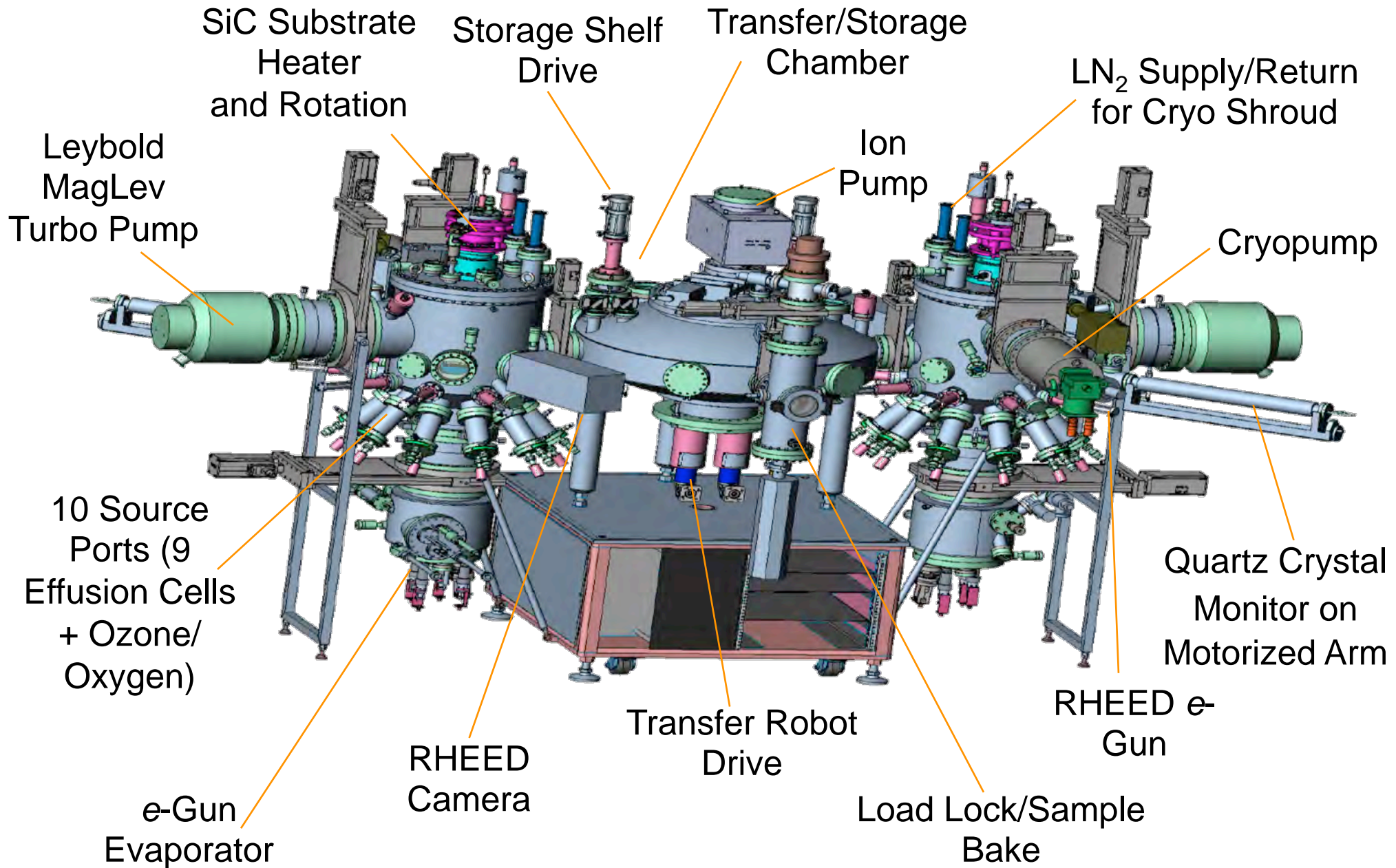


# Oxide MBE + ARPES

*Collaboration with Kyle Shen (Cornell, Physics)*



# Automated Veeco GEN10 Oxide MBE



# Sources for Oxide MBE



- **Effusion cell**  
(resistively heated thermal evaporators, up to 2000 °C), material in crucible

- **Ti-ball source**  
titanium sphere with resistive heater inside



- **e-gun evaporator**  
for extremely low vapor pressure materials (W, Ru, etc.)



# Effusion Cell Temperatures

												25 °C (gas)					Noble
												100 – 750 °C					
												750 – 1350 °C					
												1350 – 2000 °C					
												> 2000 °C ( <i>e</i> -beam)					
IA												III A	IV A	V A	VIA	VII A	
<b>H</b>												<b>B</b>	<b>C</b>	<b>N</b>	<b>O</b>	<b>F</b>	<b>He</b>
	IIA																
<b>Li</b>	<b>Be</b>											<b>Al</b>	<b>Si</b>	<b>P</b>	<b>S</b>	<b>Cl</b>	<b>Ar</b>
<b>Na</b>	<b>Mg</b>	IIIB	IVB	VB	VIB	VII B	VIII B		IB	IIB							
<b>K</b>	<b>Ca</b>	<b>Sc</b>	<b>Ti</b>	<b>V</b>	<b>Cr</b>	<b>Mn</b>	<b>Fe</b>	<b>Co</b>	<b>Ni</b>	<b>Cu</b>	<b>Zn</b>	<b>Ga</b>	<b>Ge</b>	<b>As</b>	<b>Se</b>	<b>Br</b>	<b>Kr</b>
<b>Rb</b>	<b>Sr</b>	<b>Y</b>	<b>Zr</b>	<b>Nb</b>	<b>Mo</b>	<b>Tc</b>	<b>Ru</b>	<b>Rh</b>	<b>Pd</b>	<b>Ag</b>	<b>Cd</b>	<b>In</b>	<b>Sn</b>	<b>Sb</b>	<b>Te</b>	<b>I</b>	<b>Xe</b>
<b>Cs</b>	<b>Ba</b>	†	<b>Hf</b>	<b>Ta</b>	<b>W</b>	<b>Re</b>	<b>Os</b>	<b>Ir</b>	<b>Pt</b>	<b>Au</b>	<b>Hg</b>	<b>Tl</b>	<b>Pb</b>	<b>Bi</b>	<b>Po</b>	<b>At</b>	<b>Rn</b>
<b>Fr</b>	<b>Ra</b>	‡	<b>Rf</b>	<b>Ha</b>	<b>Sg</b>	<b>Ns</b>	<b>Hs</b>	<b>Mt</b>									

†	<b>La</b>	<b>Ce</b>	<b>Pr</b>	<b>Nd</b>	<b>Pm</b>	<b>Sm</b>	<b>Eu</b>	<b>Gd</b>	<b>Tb</b>	<b>Dy</b>	<b>Ho</b>	<b>Er</b>	<b>Tm</b>	<b>Yb</b>	<b>Lu</b>
‡	<b>Ac</b>	<b>Th</b>	<b>Pa</b>	<b>U</b>	<b>Np</b>	<b>Pu</b>	<b>Am</b>	<b>Cm</b>	<b>Bk</b>	<b>Cf</b>	<b>Es</b>	<b>Fm</b>	<b>Md</b>	<b>No</b>	<b>Lr</b>

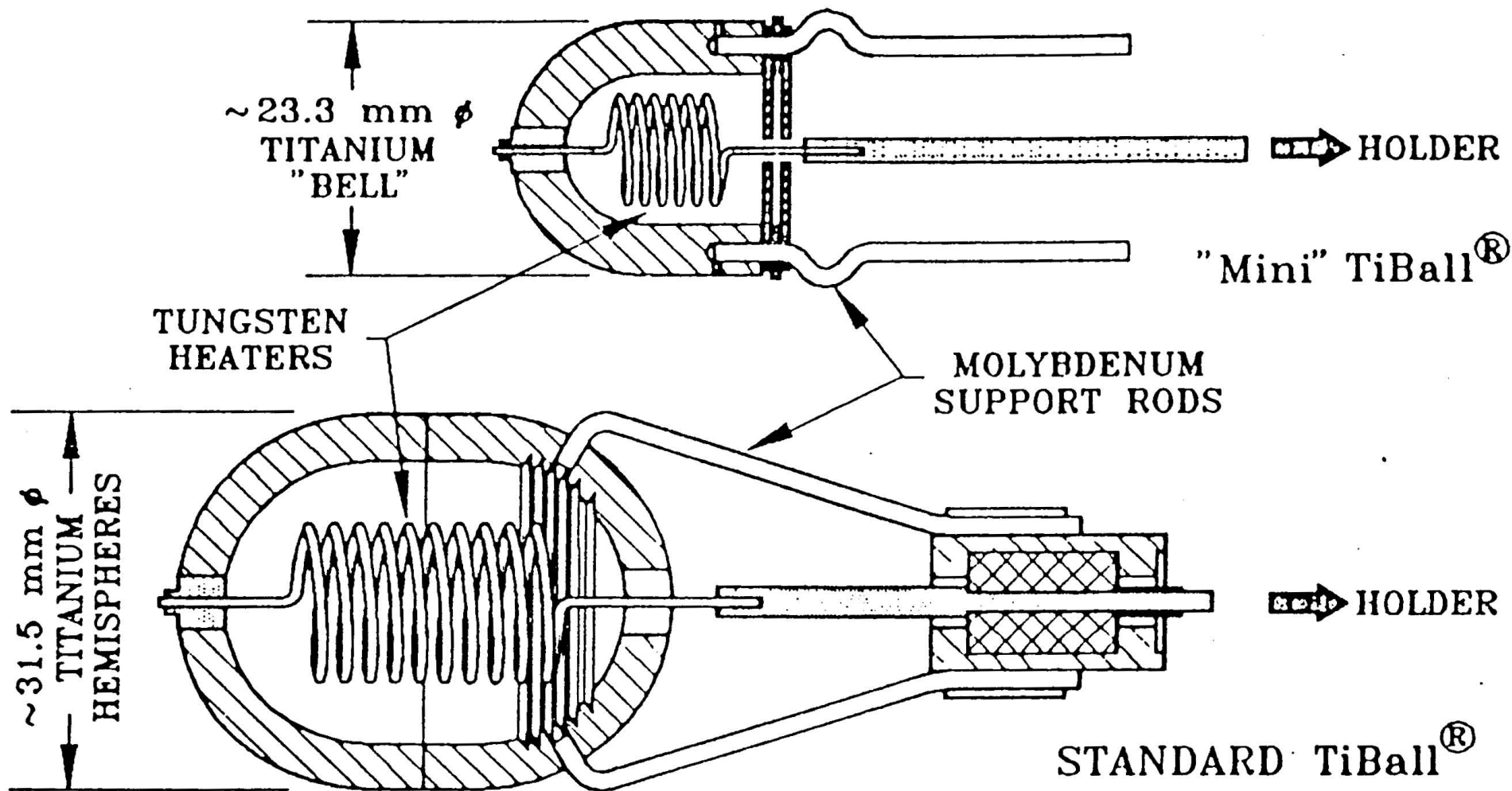


Figure 3.4.4. Two varieties of radiantly heated sublimation sources; the TiBall<sup>®</sup>, developed by Harra and Snouse<sup>(42)</sup> and a smaller, radiantly heated source developed by Welch.

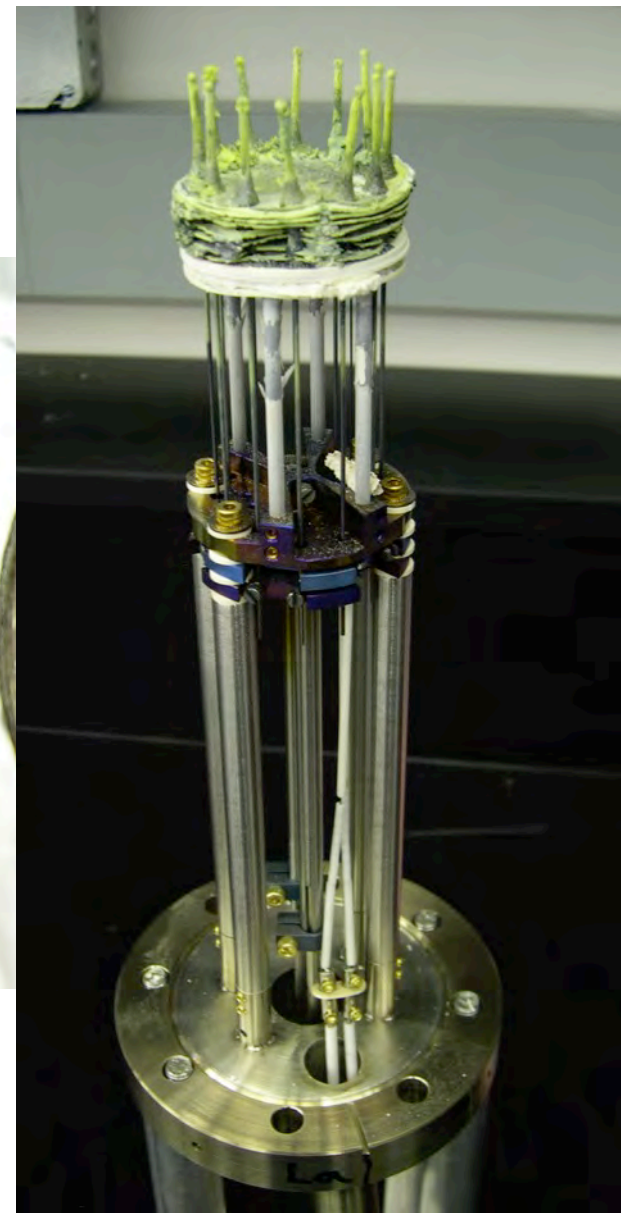
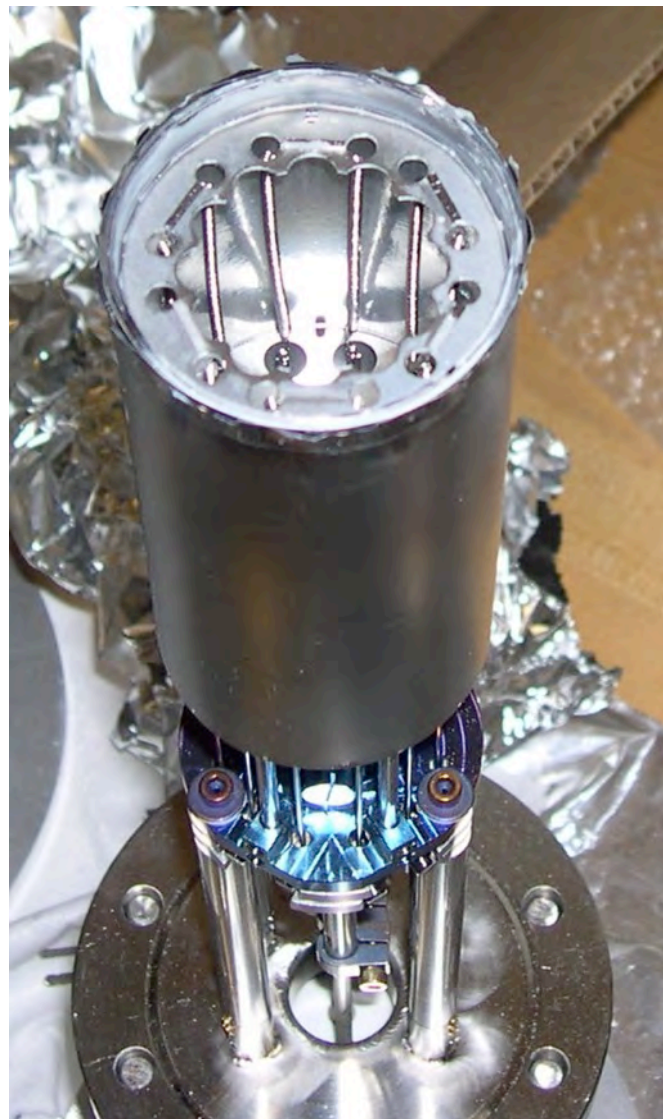
## CAPTURE PUMPING TECHNOLOGY

*An Introduction*

KIMO M. WELCH  
Brookhaven National Laboratory, Upton, NY, USA



# MBE Effusion Cells



# Arrhenius Plot of Vapor Pressure

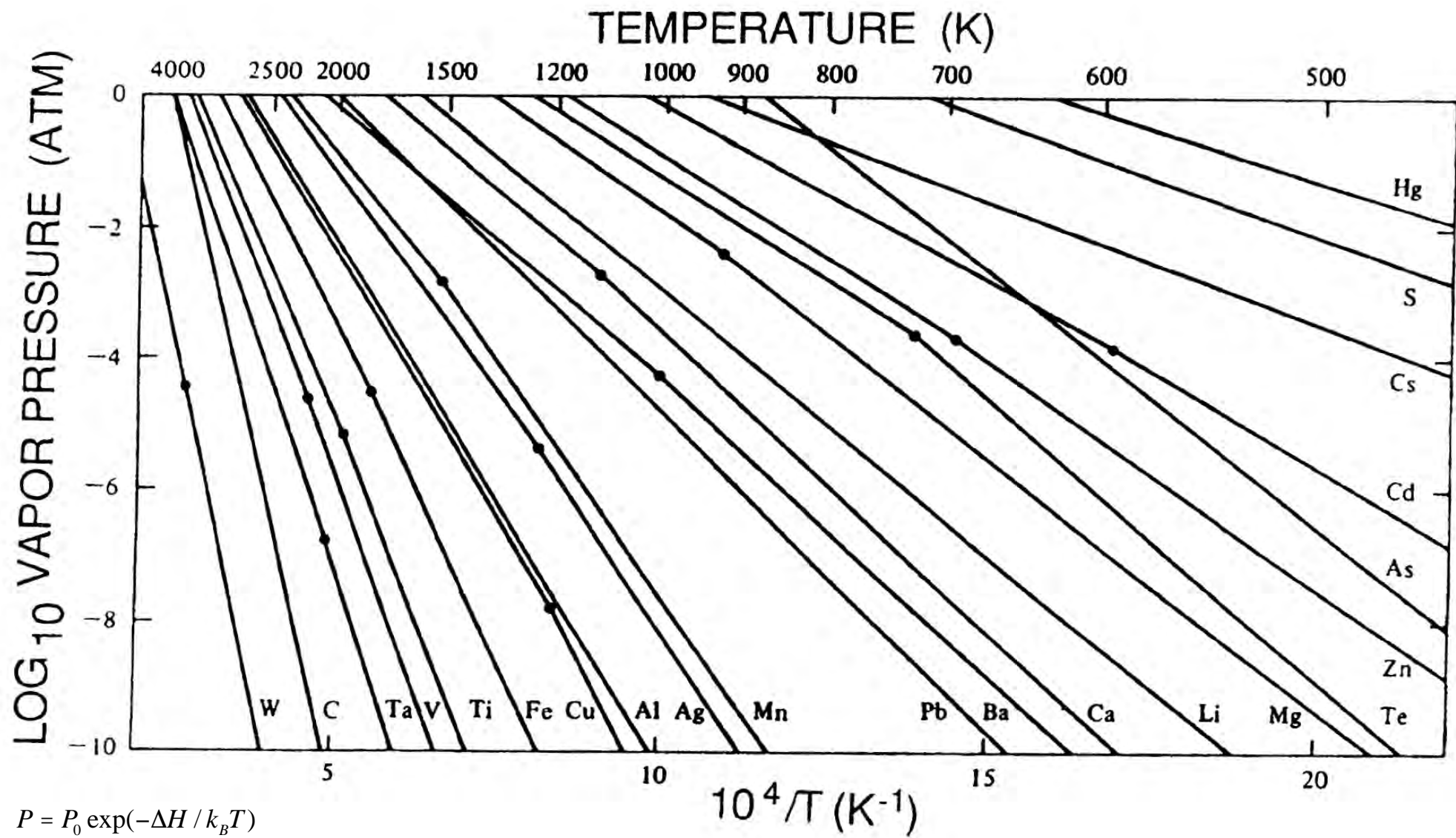
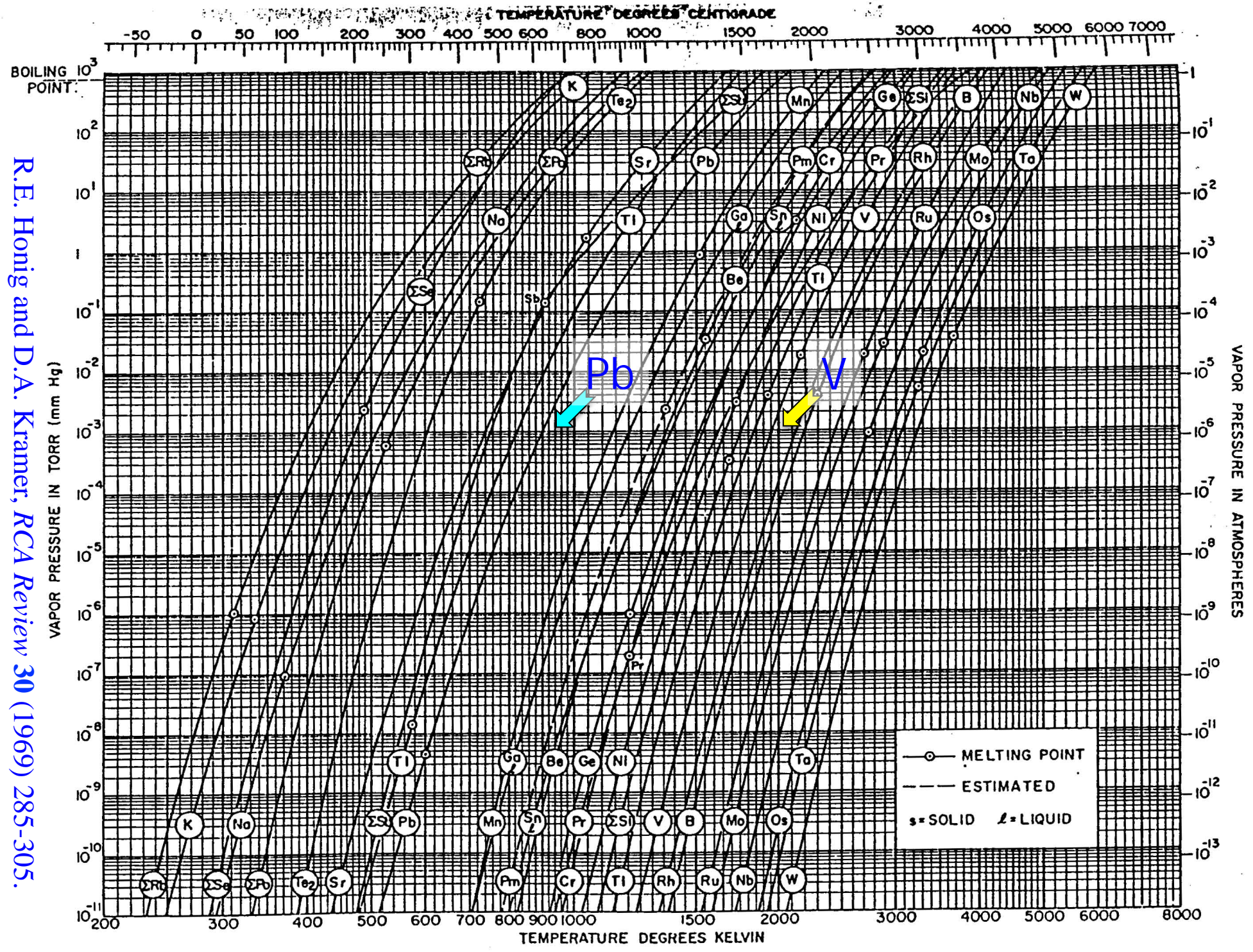


Fig. 1B—Vapor pressure data for the solid and liquid elements.

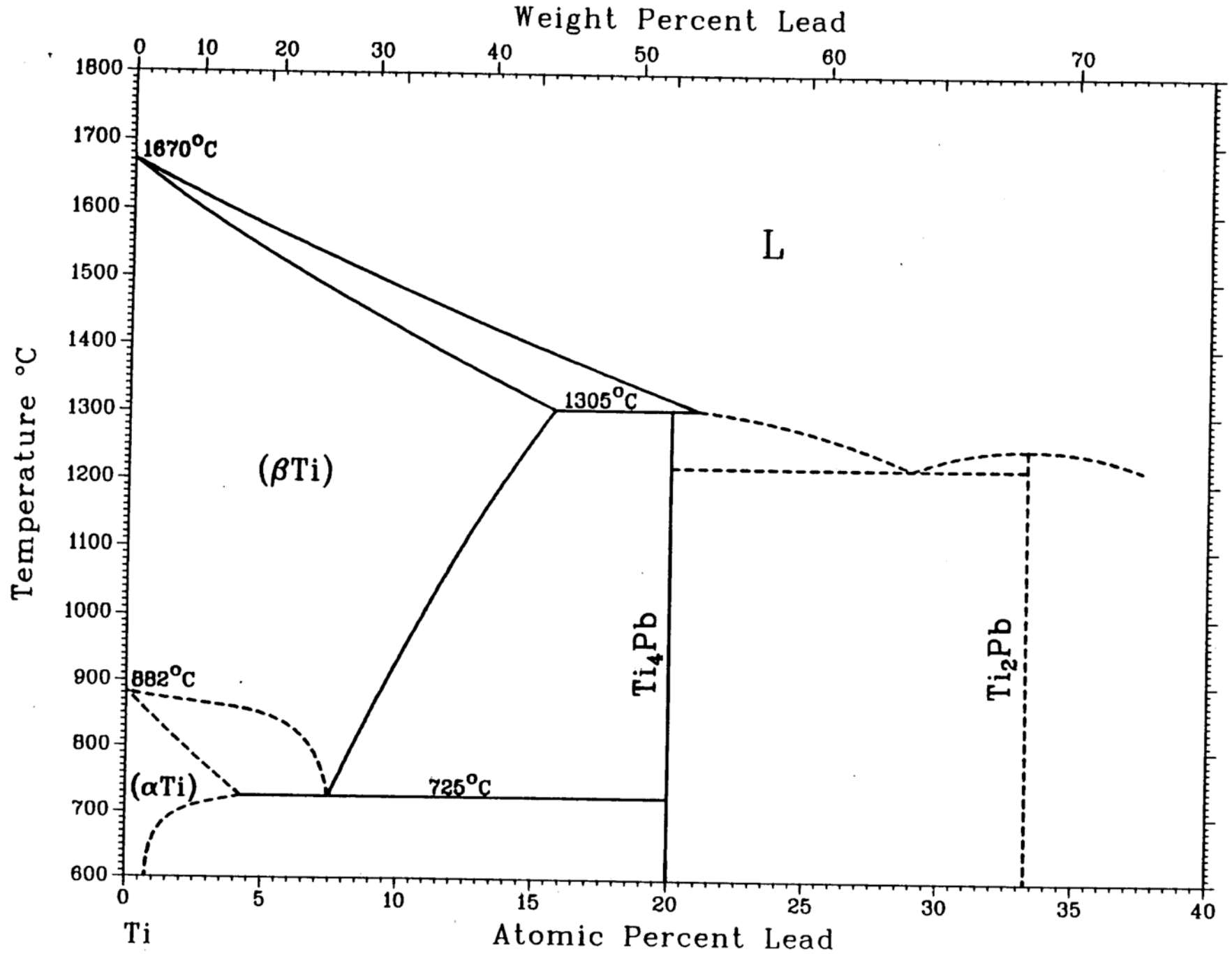
R.E. Honig and D.A. Kramer, *RCA Review* 30 (1969) 285-305.





# Ti-Pb Phase Diagram

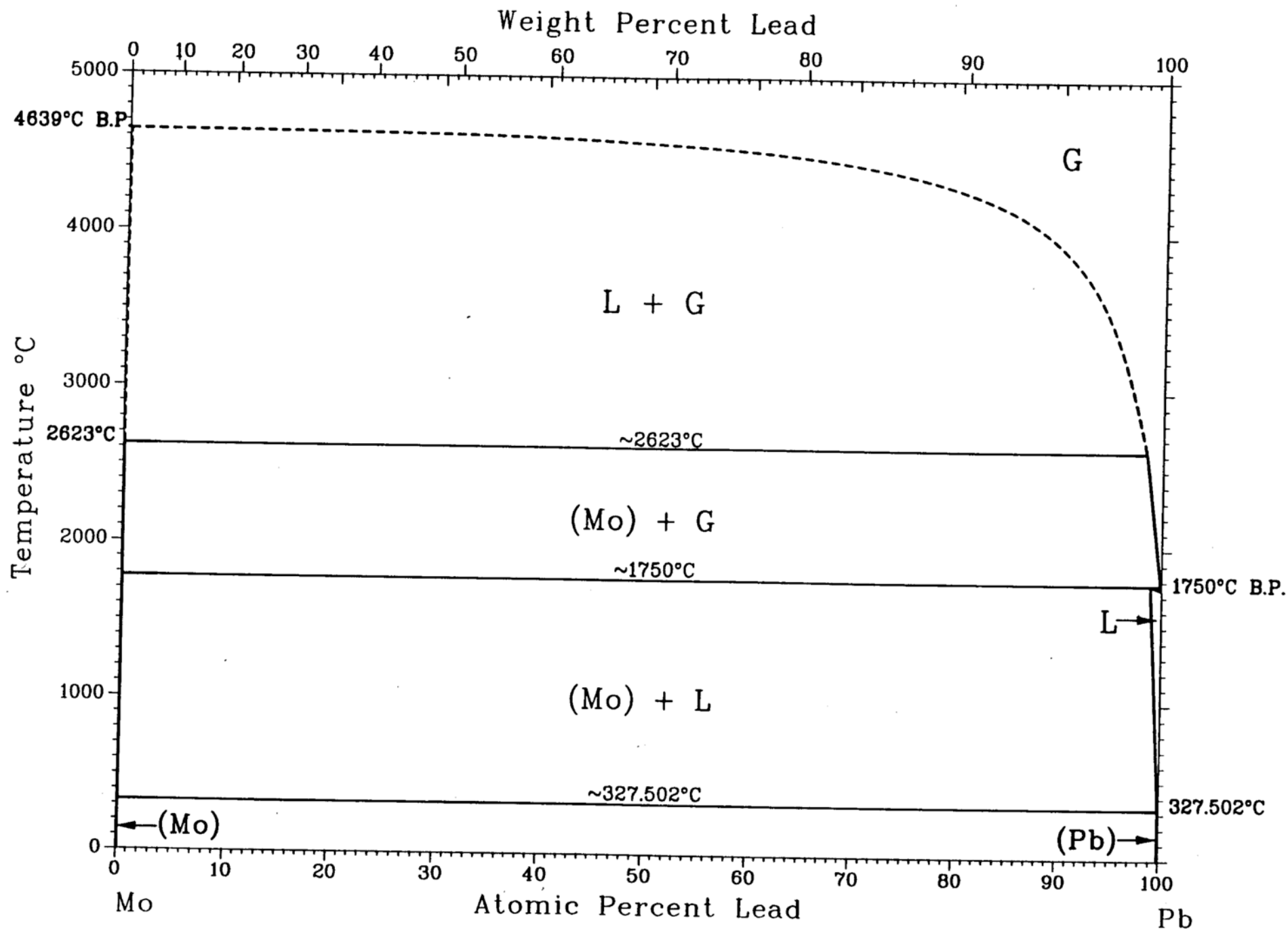
*Binary Alloy Phase Diagrams,*  
edited by T.B. Massalski (ASM International, 1990).





# Mo-Pb Phase Diagram

*Binary Alloy Phase Diagrams,*  
edited by T.B. Massalski (ASM International, 1990).



# Pb-W (Lead-Tungsten)

S.V. Nagender Naidu and P. Rama Rao



No phase diagram is available for the Pb-W system. The solubility of W in liquid Pb is less than 0.1 at.% W. No intermediate phases exist in the system.

[19Ino] claimed to have determined the solidification temperature of alloys containing up to 30 at.% W at 1300 °C; however, no further details are available in this regard. The findings of [19Ino] are not accepted here because there is no confirming evidence in any of the later experimental investiga-

## Pb-W Crystal Structure Data

Phase	Composition, at.% W	Pearson symbol	Space group	Struktur- bericht designation	Prototype
(Pb) .....	~0	<i>cF4</i>	<i>Fm<math>\bar{3}m</math></i>	A1	Cu
(W) .....	~100	<i>cI2</i>	<i>Im<math>\bar{3}m</math></i>	A2	W

tions as to the alloy formation.

**19Ino:** S. Inouye, *Mem. Coll. Sci. Kyoto Univ.*, 4, 43-46 (1919) in German.

To be published in *Phase Diagrams of Bi-*

*nary Tungsten Alloys*, 1991. Complete evaluation contains 1 table and 11 references.

# What crucible would you use for Pb in MBE?

---

(a) Ti

(b) Mo

(c) W

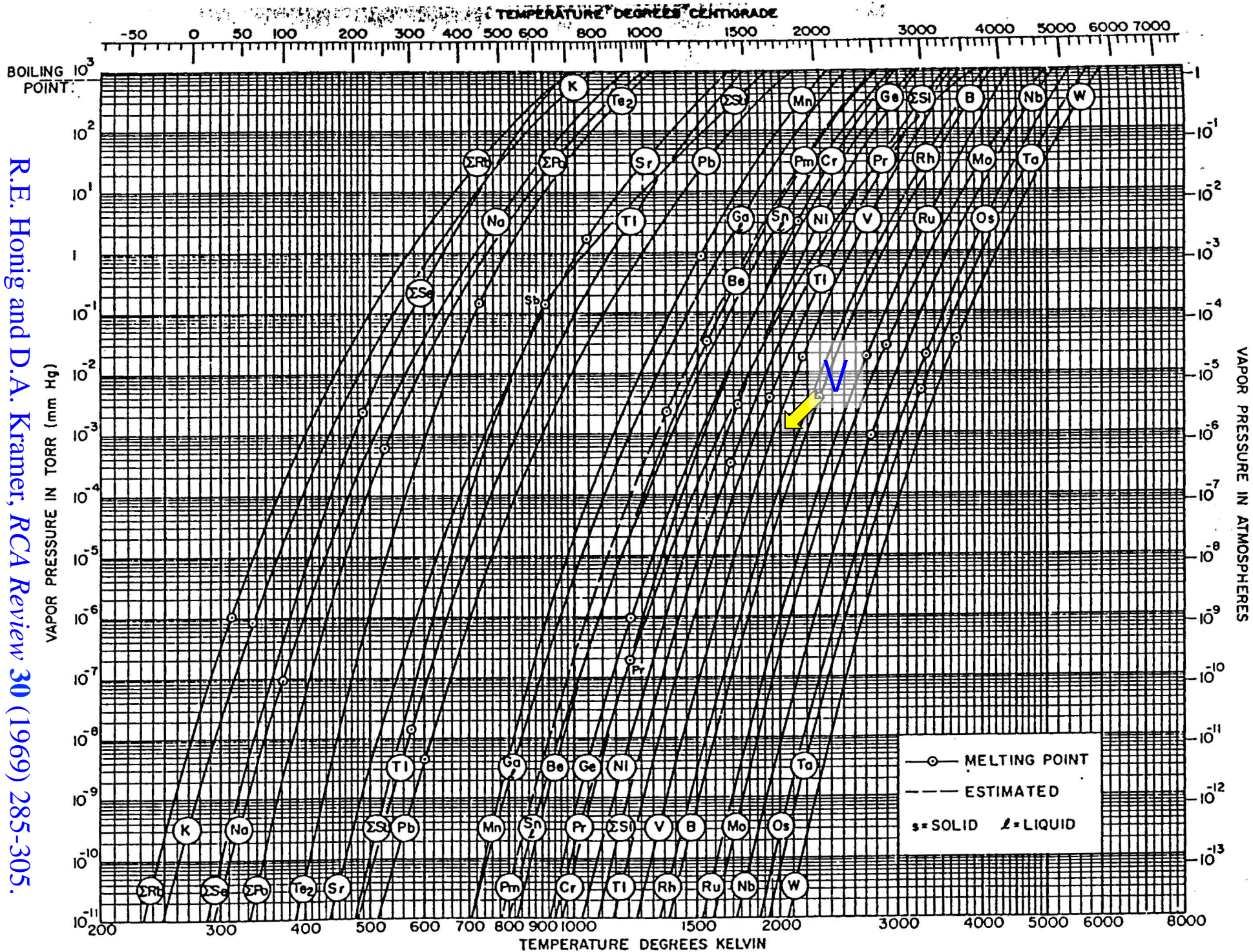
Material	Symbol	MP (° C)	S/D	g/cm <sup>3</sup>	Temp. (° C) for Given Vap. Press. (Torr)			Evaporation Techniques					Sputter	Comments
					10 <sup>-8</sup>	10 <sup>-6</sup>	10 <sup>-4</sup>	E-Beam	Boat	Coil	Basket	Crucible		
Kanthal	FeCrAl	—	—	7.1	—	—	—	—	W	W	W	—	DC, RF	—
Lanthanum	La	921	—	6.15	990	1,212	1,388	Ex	W, Ta	—	—	Al <sub>2</sub> O <sub>3</sub>	RF	Films will burn in air if scraped
Lanthanum Boride	LaB <sub>6</sub>	2,210	D	2.61	—	—	—	G	—	—	—	—	RF	—
Lanthanum Bromide	LaBr <sub>3</sub>	783	—	5.06	—	—	—	—	—	—	Ta	—	RF	n=1.94. Hygroscopic
Lanthanum Fluoride	LaF <sub>3</sub>	1,490	S	~6.0	—	—	900	G	Ta, Mo	—	Ta	—	RF	No decomposition. n ~1.6
Lanthanum Oxide	La <sub>2</sub> O <sub>3</sub>	2,307	—	6.51	—	—	1,400	G	W, Ta	—	—	—	RF	Loses oxygen. n~1.73
Lead	Pb	328	—	11.34	342	427	497	Ex	W, Mo	W	W, Ta	Al <sub>2</sub> O <sub>3</sub> , Q	DC,	RF Toxic
Lead Bromide	PbBr <sub>2</sub>	373	—	6.66	—	—	~300	—	—	—	—	—	—	—
Lead Chloride	PbCl <sub>2</sub>	501	—	5.85	—	—	~325	—	Pt	—	—	Al <sub>2</sub> O <sub>3</sub>	RF	Little decomposition
Lead Fluoride	PbF <sub>2</sub>	855	S	8.24	—	—	~400	—	W, Pt, Mo	—	—	BeO	RF	n = 1.75
Lead Iodide	PbI <sub>2</sub>	402	—	6.16	—	—	~500	—	Pt	—	—	Q	—	—
Lead Oxide	PbO	886	—	9.53	—	—	~550	—	Pt	—	—	Q, Al <sub>2</sub> O <sub>3</sub>	RF-R	No decomposition. n ~2.6
Lead Selenide	PbSe	1,065	S	8.10	—	—	~500	—	W, Mo	—	W	Gr, Al <sub>2</sub> O <sub>3</sub>	RF	—
Lead Stannate	PbSnO <sub>3</sub>	1,115	—	8.1	670	780	905	P	Pt	—	Pt	Al <sub>2</sub> O <sub>3</sub>	RF	Disproportionates
Lead Sulfide	PbS	1,114	S	7.5	—	—	500	—	W	—	W, Mo	Q, Al <sub>2</sub> O <sub>3</sub>	RF	Little decomposition. n = 3.92
Lead Telluride	PbTe	917	—	8.16	780	910	1,050	—	Mo, Pt, Ta	—	—	Al <sub>2</sub> O <sub>3</sub> , Gr	RF	Vapors toxic. Deposits aretellurium rich. Sputtering preferred or co-evaporate from two sources
Lead Titanate	PbTiO <sub>3</sub>	—	—	7.52	—	—	—	—	Ta	—	—	—	RF	—
Lithium	Li	181	—	0.53	227	307	407	G	Ta, SS	—	—	Al <sub>2</sub> O <sub>3</sub> , BeO	—	Metal reacts quickly in air
Lithium Bromide	LiBr	550	—	3.46	—	—	~500	—	Ni	—	—	—	RF	n = 1.78
Lithium Chloride	LiCl	605	—	2.07	—	—	400	—	Ni	—	—	—	RF	Preheat gently to outgas. n = 1.66
Lithium Fluoride	LiF	845	—	2.64	875	1,020	1,180	G	Ni, Ta, Mo, W	—	—	Al <sub>2</sub> O <sub>3</sub>	RF	Rate control important for optical films. Preheat gently to outgas. n = 1.39
Lithium Iodide	LiI	449	—	4.08	—	—	400	—	Mo, W	—	—	—	RF	n = 1.96
Lithium Oxide	Li <sub>2</sub> O	>1,700	—	2.01	—	—	850	—	Pt, Ir	—	—	—	RF	n = 1.64
Lutetium	Lu	1,663	—	9.84	—	—	1,300	Ex	Ta	—	—	Al <sub>2</sub> O <sub>3</sub>	RF, DC	—
Lutetium Oxide	Lu <sub>2</sub> O <sub>3</sub>	—	—	9.42	—	—	1,400	—	Ir	—	—	—	RF	Decomposes

**Key of Symbols:** \* influenced by composition; \*\* Cr-plated rod or strip; \*\*\*all metals alumina coated; **C** = carbon; **Gr** = graphite; **Q** = quartz; **Incl** = Inconel; **VC** = vitreous carbon; **SS** = stainless steel; **Ex** = excellent; **G** = good; **F** = fair; **P** = poor; **S** = sublimes; **D** = decomposes; **RF** = RF sputtering is effective; **RF-R** = reactive RF sputter is effective; **DC** = DC sputtering is effective; **DC-R** = reactive DC sputtering is effective



Fig. 1B—Vapor pressure data for the solid and liquid elements.

R.E. Honig and D.A. Kramer, *RCA Review* 30 (1969) 285-305.

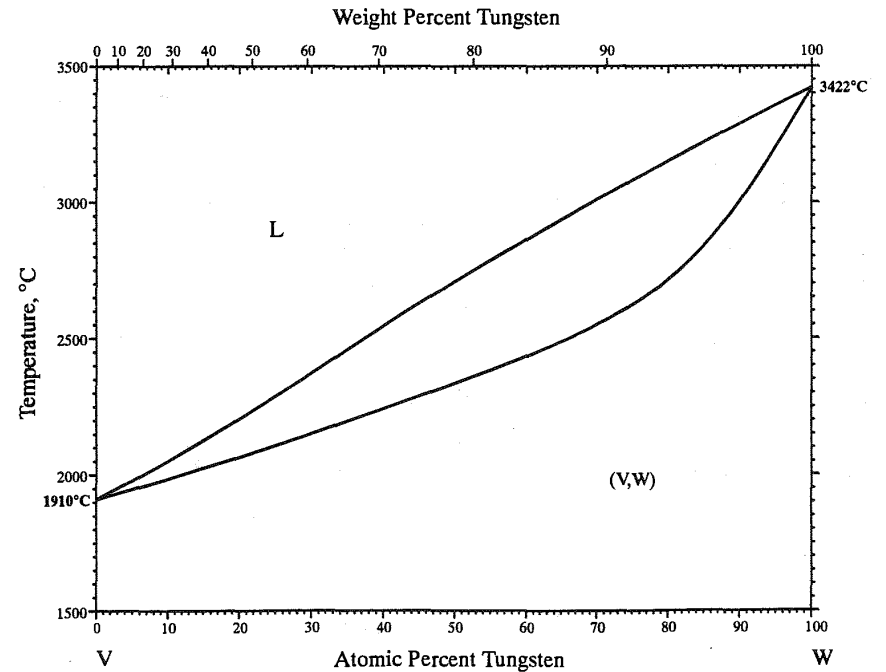
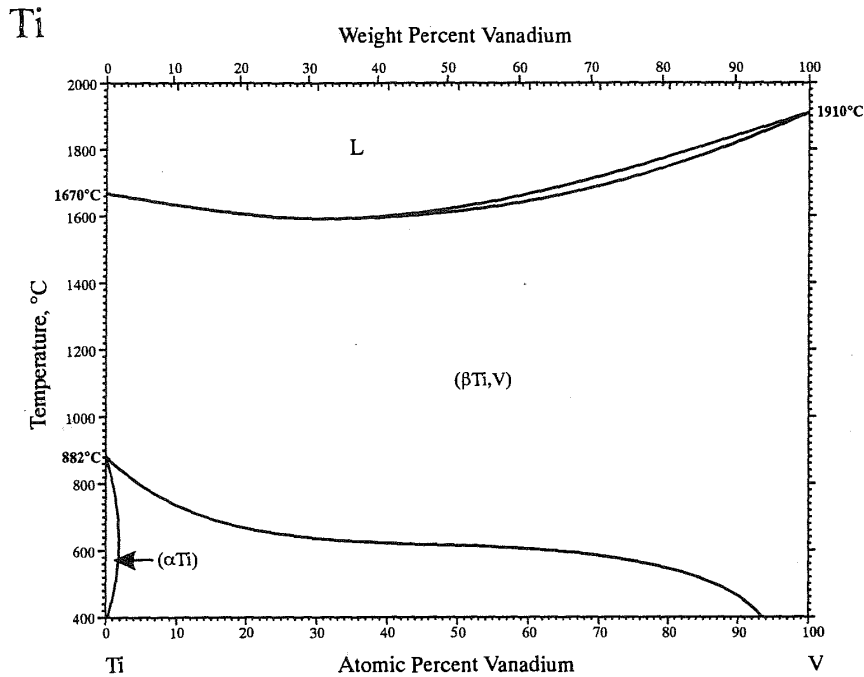
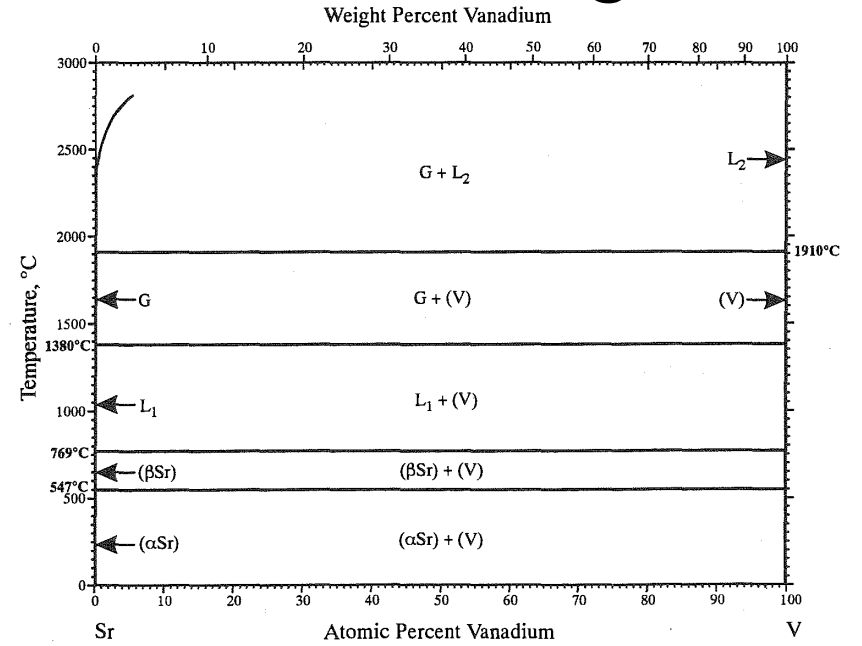
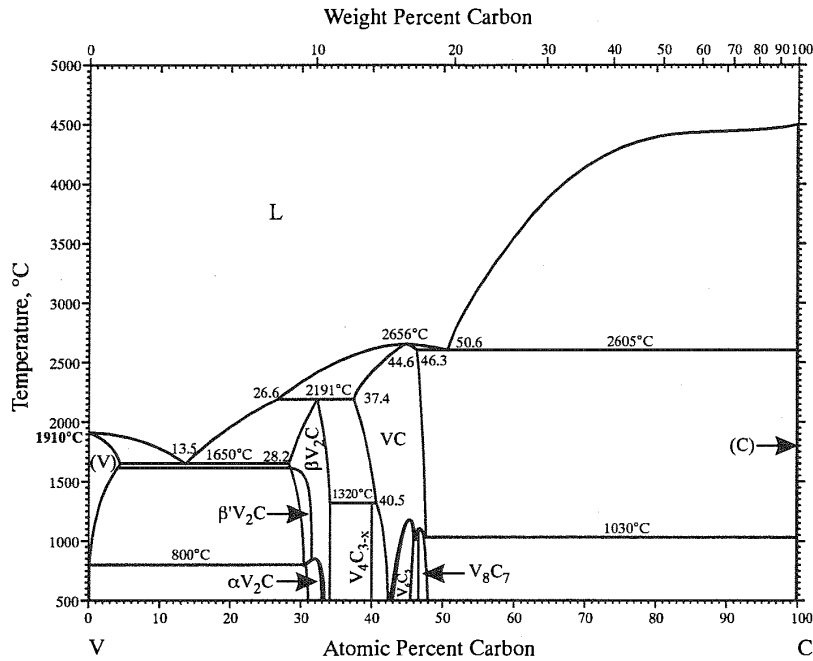


Material	Symbol	MP (° C)	S/D	g/cm <sup>3</sup>	Temp. (° C) for Given			Evaporation Techniques					Sputter	Comments	
					Vap. Press. (Torr)			E-Beam	Boat	Thermal Sources					Crucible
					10 <sup>-8</sup>	10 <sup>-6</sup>	10 <sup>-4</sup>			Coil	Basket				
Thulium	Tm	1,545	S	9.32	461	554	680	G	Ta	—	—	Al <sub>2</sub> O <sub>3</sub>	DC	—	
Thulium Oxide	Tm <sub>2</sub> O <sub>3</sub>	—	—	8.90	—	—	1,500	—	Ir	—	—	—	RF	Decomposes	
Tin	Sn	232	—	7.28	682	807	997	Ex	Mo	W	W	Al <sub>2</sub> O <sub>3</sub>	DC, RF	Wets molybdenum Use tantalum liner in E-beam guns	
Tin Oxide	SnO <sub>2</sub>	1,630	S	6.95	—	—	~1,000	Ex	W	W	W	Q, Al <sub>2</sub> O <sub>3</sub>	RF, RF-R	Films from tungsten are oxygen deficient, oxidize in air. n = 2.0	
Tin Selenide	SnSe	861	—	6.18	—	—	~400	G	—	—	—	Q	RF	—	
Tin Sulfide	SnS	882	—	5.22	—	—	~450	—	—	—	—	Q	RF	—	
Tin Telluride	SnTe	780	D	6.48	—	—	~450	—	—	—	—	Q	RF	—	
Titanium	Ti	1,660	—	4.5	1,067	1,235	1,453	Ex	W	—	—	TiC	DC, RF	Alloys with refractory metals; evolves gas on first heating	
Titanium Boride	TiB <sub>2</sub>	2,900	—	4.50	—	—	—	P	—	—	—	—	RF, DC	—	
Titanium Carbide	TiC	3,140	—	4.93	—	—	~2,300	—	—	—	—	—	RF, DC	—	
Titanium Nitride	TiN	2,930	—	5.22	—	—	—	G	Mo	—	—	—	RF, RF-R, DC	Sputtering preferred. Decomposes with thermal evaporation	
Titanium (II) Oxide	TiO	1,750	—	4.93	—	—	~1,500	G	W, Mo	—	—	VC	RF	Preheat gently to outgas. n = 2.2	
Titanium (III) Oxide	Ti <sub>2</sub> O <sub>3</sub>	2,130	D	4.6	—	—	—	G	W	—	—	—	RF	Decomposes	
Titanium (IV) Oxide	TiO <sub>2</sub>	1,830	—	4.26	—	—	~1,300	F	W, Mo	—	W	—	RF, RF-R	Suboxide, must be reoxidized to rutile. Tantalum reduces TiO <sub>2</sub> to TiO and titanium. n = 2.616, 2.903	
Tungsten	W	3,410	—	19.35	2,117	2,407	2,757	G	—	—	—	—	RF, DC	Forms volatile oxides. Films hard and adherent	
Tungsten Boride	WB <sub>2</sub>	~2,900	—	10.77	—	—	—	P	—	—	—	—	RF	—	
Tungsten Carbide	W <sub>2</sub> C	2,860	—	17.15	1,480	1,720	2,120	Ex	C	—	—	—	RF, DC	—	
Tungsten Disulfide	WS <sub>2</sub>	1,250	D	7.5	—	—	—	—	—	—	—	—	RF	—	
Tungsten Oxide	WO <sub>3</sub>	1,473	S	7.16	—	—	980	G	W, Pt	—	—	—	RF-R	Preheat gently to outgas. Tungsten reduces oxide slightly. n = 1.68	
Tungsten Selenide	WSe <sub>2</sub>	—	—	9.0	—	—	—	—	—	—	—	—	RF	—	
Tungsten Silicide	WSi <sub>2</sub>	>900	—	9.4	—	—	—	—	—	—	—	—	RF, DC	—	
Tungsten Telluride	WTe <sub>3</sub>	—	—	9.49	—	—	—	—	—	—	—	Q	RF	—	
Uranium	U	1,132	—	19.05	1,132	1,327	1,582	G	Mo, W	W	W	—	—	Films oxidize	
Uranium Carbide	UC <sub>2</sub>	2,350	—	11.28	—	—	2,100	—	—	—	—	C	RF	Decomposes	
Uranium Fluoride	UF <sub>4</sub>	960	—	6.70	—	—	300	—	Ni	—	—	—	RF	—	
Uranium (III) Oxide	U <sub>2</sub> O <sub>3</sub>	1,300	D	8.30	—	—	—	—	W	—	W	—	RF-R	Disproportionates at 1,300° C to UO <sub>2</sub>	
Uranium (IV) Oxide	UO <sub>2</sub>	2,878	—	10.96	—	—	—	—	W	—	W	—	RF	Tantalum causes decomposition	
Uranium Phosphide	UP <sub>2</sub>	—	—	8.57	—	—	1,200	—	Ta	—	—	—	RF	Decomposes	
Uranium (II) Sulfide	US	>2,000	—	10.87	—	—	—	—	—	—	—	—	—	—	
Uranium (IV) Sulfide	US <sub>2</sub>	>1,100	—	7.96	—	—	—	—	W	—	—	—	RF	Slight decomposition	
Vanadium	V	1,890	—	5.96	1,162	1,332	1,547	Ex	W, Mo	—	—	—	DC, RF	Wets molybdenum. E-beam-evaporated films preferred. n = 3.03	
Vanadium Boride	VB <sub>2</sub>	2,400	—	5.10	—	—	—	—	—	—	—	—	RF, DC	—	
Vanadium Carbide	VC	2,810	—	5.77	—	—	~1,800	—	—	—	—	—	RF, DC	—	
Vanadium Nitride	VN	2,320	—	6.13	—	—	—	—	—	—	—	—	RF, RF-R, DC	—	
Vanadium (IV) Oxide	VO <sub>2</sub>	1,967	S	4.34	—	—	~575	—	—	—	—	—	RF, RF-R	Sputtering preferred.	
Vanadium (V) Oxide	V <sub>2</sub> O <sub>5</sub>	690	D	3.36	—	—	~500	—	—	—	—	Q	RF	n = 1.46, 1.52, 1.76	
Vanadium Silicide	VSi <sub>2</sub>	1,700	—	4.42	—	—	—	—	—	—	—	—	RF	—	
Ytterbium	Yb	819	S	6.96	520	590	690	G	Ta	—	—	—	DC, RF	—	
Ytterbium Fluoride	YbF <sub>3</sub>	1,157	—	—	—	—	~800	—	Mo	—	—	—	RF	—	
Ytterbium Oxide	Yb <sub>2</sub> O <sub>3</sub>	2,346	S	9.17	—	—	~1,500	—	Ir	—	—	—	RF, RF-R	Loses oxygen	

**Key of Symbols:** \* influenced by composition; \*\* Cr-plated rod or strip; \*\*\*all metals alumina coated; **C** = carbon; **Gr** = graphite; **Q** = quartz; **Incl** = Inconel; **VC** = vitreous carbon; **SS** = stainless steel; **Ex** = excellent; **G** = good; **F** = fair; **P** = poor; **S** = sublimates; **D** = decomposes; **RF** = RF sputtering is effective; **RF-R** = reactive RF sputter is effective; **DC** = DC sputtering is effective; **DC-R** = reactive DC sputtering is effective

# C-V, Sr-V, Ti-V, and W-V Phase Diagrams

Desk Handbook: Phase Diagrams for Binary Alloys, edited by H. Okamoto (ASM International, 2000).



# What crucible would you use for V in MBE?

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(a) C

(b) Sr

(c) Ti

(d) W